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Facile preparation and characterization of injectable self-antibacterial gelatin/carrageenan/bacterial cellulose hydrogel scaffolds for wound healing application

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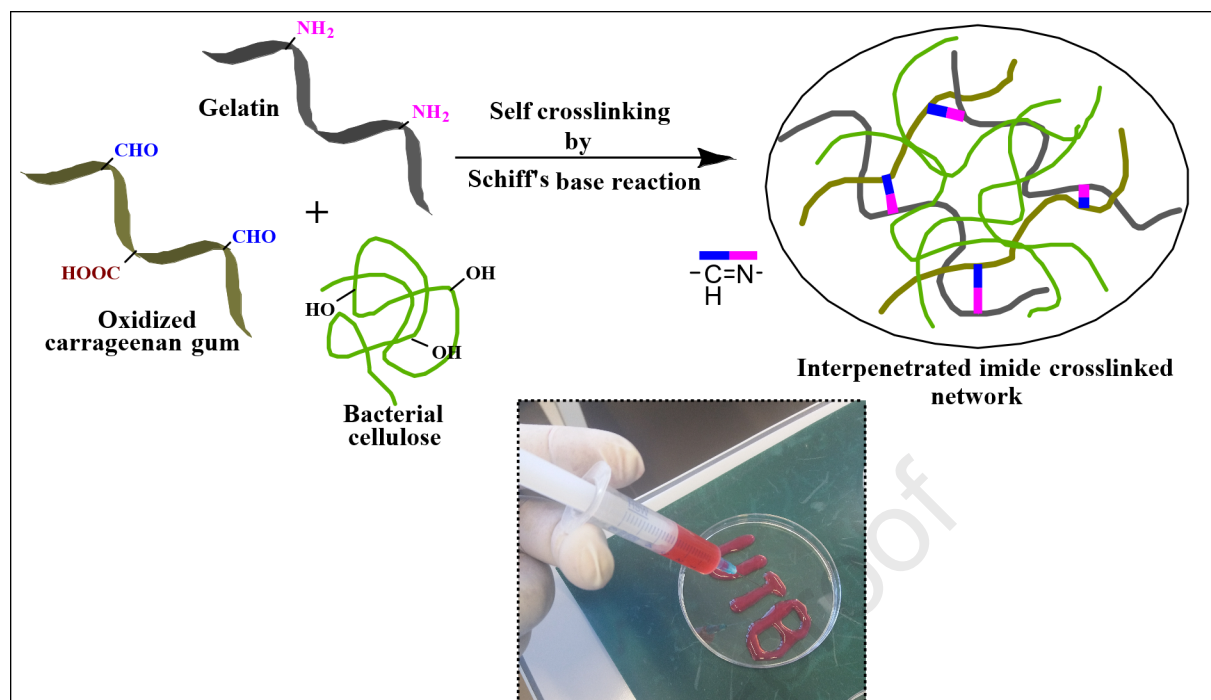
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Author contributions:

This study conceptualization and methodology was performed by Fahanwi Asabuwa Ngwabebhoh. Formal analysis was done by Fahanwi Asabuwa Ngwabebhoh, Rahul Patwa and Oyunchimeg Zandraa. Investigation and data curation were carried out by Fahanwi Asabuwa Ngwabebhoh, Rahul Patwa and Oyunchimeg Zandraa. Writing-original draft preparation was done by Fahanwi Asabuwa Ngwabebhoh. The paper was reviewed, edited and supervised by Nabanita Saha and Petr Saha.



1 **Research Paper**2 **Facile preparation and characterization of injectable self-antibacterial**
3 **gelatin/carrageenan/bacterial cellulose hydrogel scaffolds for wound**
4 **healing application**5 **Fahanwi Asabuwa Ngwabebhoh^{1,2*}, Rahul Patwa^{1,2}, Oyunchimeg Zandraa^{1,2}, Nabanita**
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10 *3685, Zlin, Czech Republic*11 ³*Faculty of Technology, Tomas Bata University in Zlin, Vavrečkova 275, 76001 Zlin, Czech Republic*12 **Abstract**

13 This study reports the preparation of self-crosslinked Schiff base gels using gelatin and
14 oxidized carrageenan gum interpenetrated with bacterial cellulose (BC) as injectable drug
15 delivery systems. The injectable gels were successfully prepared at body temperature upon
16 blending with BC and loaded with bovine serum albumin (as the model drug) to produce
17 scaffolds. The gel scaffolds were characterized via rheological, FTIR, SEM, XRD, TGA and
18 mechanical compression analysis. Gelation kinetics of gels as well as swelling, *in vitro*
19 degradation and drug release kinetics of gel scaffolds were examined. Results showed that the
20 incorporation of BC to the gel system considerably improved mechanical integrity with
21 remarkable rheological shear-thinning properties. Maximum *in vitro* cumulative drug release
22 from gel scaffolds was determined as $84.01 \pm 3.66\%$ within the studied time interval of 168
23 h. Further analysis showed that the prepared gel scaffolds possess self-antibacterial properties
24 with growth inhibition capacity against *E. coli*, *S. aureus*, and *K. pneumonia*. *In vitro* cell
25 cytotoxicity was also performed by MTT assay and results depicted >80% cell viability,
26 which indicates the gel scaffolds are cytocompatible. In conclusion, this paper presents a
27 facile approach to fabricate all-natural crosslinked injectable self-antibacterial gels systems
28 with prospective potential application in wound dressing and tissue regeneration.

29

30

31 **Keywords:** Hydrogel, Gelatin, Bacterial cellulose, Drug delivery, Antibacterial activity

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39

40 1. Introduction

41 Wound healing as a complex and coordinated process which has been studied extensively
42 towards enhancing healing by using beneficial wound dressing materials. Recently, different
43 wound dressing materials, such as films, nanoparticles, hydrocolloids, and hydrogels have
44 been commercialized and are studied extensively [1, 2]. These dressing materials help
45 provide a moist environment for cell regeneration, protect the wounds from bacterial
46 infections and absorb excess exudate [3]. Amongst these materials, hydrogels have proven
47 attractive features by acting as a physical barrier, fluid absorbent and provide moist scaffold
48 for tissue regeneration. However, most hydrogels have demonstrated to possess low
49 mechanical stability and may require a secondary dressing to adhere on skin tissue [4, 5].

50 Injectable hydrogels have demonstrated to be attractive wound dressing material as they
51 are less invasive, and can be used for the local delivery of therapeutic agents and cost-
52 effective as it can replace surgical procedures [6]. Over the years, several injectable
53 hydrogels have been developed using synthetic polymers such as poly(N-isopropyl
54 acrylamide), but of which many are non-biodegradable and may lead to local inflammation at
55 infected sites [7]. Naturally-derived polymers or polysaccharides provide the required
56 friendly extracellular matrix for cell attachment and proliferation, which serve as suitable
57 drug carriers and tissue regeneration scaffolds [8, 9]. Gelatin a collagen-derived polymer is
58 made up of peptide sequences that aid in the recognition of integrin receptors in cells, which
59 are vital for cell adhesion in wound healing. However, pristine gelatin possesses low gelling
60 temperature ($< 30\text{ }^{\circ}\text{C}$) that hinders its usage at human body temperature ($37\text{ }^{\circ}\text{C}$) [10].
61 Therefore, procedures including physical blending and chemical modification have been
62 developed to enhance its gelling conditions. This includes chemical grafting with thiol and
63 methacryloyl groups, use of crosslinking agents such as glutaraldehyde, epichlorohydrin,
64 genipin, or aldehyde-oxidized molecules as well as blending with other polymers to form gels
65 [11-14]. Amongst all, the use of aldehyde-oxidized polymers to form hydrogels via self-
66 crosslinking mechanisms have demonstrated to be very efficient since the toxicity effect of
67 some crosslinking agents are avoided, which to some extent greatly enhances the
68 performance of the hydrogels formed. So far, polysaccharide aldehyde-oxidized moieties
69 have been widely synthesized via oxidation at C2 and C3 position to form excellent
70 polymeric dialdehyde crosslinkers [15]. Carrageenan gum (CG) a linear sulphated
71 polysaccharide mainly extracted from red seaweeds has been widely modified via oxidation
72 using different oxidizing agents such as sodium periodate, sodium hypochlorite, 2,2,6,6-

73 tetramethylpyperidine-1-oxy (TEMPO) and hydrogen peroxide to produce a biocompatible
74 dialdehyde polysaccharide crosslinker [16-19]. Amongst all, hydrogen peroxide has proven to
75 be the most environmentally friendly since it inevitably decomposes to oxygen and water at
76 the end of the oxidation reaction with no harmful by-product [20]. Therefore, hydrogen
77 peroxide was applied in the present study as the oxidizing agent in the preparation of
78 oxidized CG.

79 In recent years, the potential of interpenetrating chemically crosslinked hydrogel systems
80 with biopolymeric fibres have attracted great research since this bestows the resultant
81 material with good mechanical integrity [21, 22]. Bacterial cellulose (BC) a linear
82 biopolymer composed of β -1,4-glucopyranose residues commonly biosynthesized using
83 *Komagataeibacter xylinus* bacteria has been use for this purpose [23]. Compared to cellulose
84 extracted from plants, BC shows superior physical and mechanical properties (such as higher
85 degree of purity, high water holding capacity, hydrophilicity and tensile strength) [24]. In
86 addition, BC is highly attractive for biomedical applications due to its non-toxic and
87 biocompatible properties [25]. For example, in wound dressing BC has proven promising due
88 to its complex three-dimensional structure that assures high tensile strength and flexibility
89 providing an adequate moist and thermal environment, which ensures gas and liquid
90 permeability [26, 27].

91 The choice of polymers and crosslinkage in the development of gels systems for
92 application particularly in the area of wound dressing/tissue regeneration is of utmost
93 importance. In recent years, polymers and gels systems with minimal toxicity to mammalian
94 cells and self-activity against microbes have received considerable attention [6, 28, 29].
95 Herein, we report the preparation of injectable gels with self-antibacterial properties formed
96 from the crosslinking of gelatin and aldehyde modified carrageenan gum and semi-
97 interpenetration with BC. To the best of our knowledge, this is the first work on such
98 crosslinked injectable gel system with self-antibacterial properties. A gelling mechanism was
99 proposed and the gelation kinetics of the prepared gels investigated. The rheological shear-
100 thinning and self-recovering capabilities as well compression modulus of elasticity of the
101 prepared gels were analysed. In addition, various characterizations were performed for the
102 obtained gel scaffolds as well as swelling and *in vitro* degradation were examined.
103 Cumulative release profile based on the release of bovine serum albumin (BSA) as the model
104 protein drug from the gel scaffold matrix was further evaluated. Also, biological performance

105 was tested using mouse embryonic fibroblast cells. Finally, antibacterial properties against *E.*
106 *coli*, *S. aureus*, and *K. pneumonia* were examined.

107 **2. Materials and methods**

108 *2.1. Materials*

109 Gelatin powder (Type A, from porcine skin) and kappa-carrageenan gum (CG) were
110 purchased from Sigma-Aldrich. Bacterial cellulose (BC) used was synthesized in our
111 laboratory as described previously [30]. 30 wt% hydrogen peroxide (H₂O₂) was purchased
112 from Pentachemicals Ltd and used as the oxidizing agent. Reagents such as ethylene glycol,
113 sodium hydroxide (NaOH), copper sulphate pentahydrate (CuSO₄ · 5H₂O), calcium acetate
114 Ca(CH₃COO)₂, phenolphthalein, and sulphuric acid (H₂SO₄, ≥95% purity) were supplied by
115 Sigma Aldrich. Lyophilized powder of lysozyme (from chicken egg white, protein ≥90%)
116 and bovine serum albumin (BSA, used as the model drug) were also purchased from Sigma
117 Aldrich. All chemicals were used without further purification. Double distilled water was use
118 for the preparation of all aqueous solutions.

119 *2.2. Synthesis of dialdehyde carrageenan gum polymer*

120 The oxidized carrageenan gum (OCG) was prepared via a H₂O₂ and CuSO₄ redox reaction
121 as described previously [31, 32]. Initially, 4 g of CG polymer was dissolved in 200 mL of
122 distilled water at 80 °C, followed by cooling to 50 °C. Into the cooled dissolved mixture, 65
123 mL of 30 wt% H₂O₂ and 10 mL of 0.05 wt% CuSO₄, changing the colour of mixture to light
124 brown. The mixture was then reacted at 50 °C for 5 h under constant stirring, which resulted
125 in a colourless solution. The obtained mixture was dialyzed for 3 days using a dialysis bag
126 (molecular weight cut-off 12kDa) to eliminate the copper ions and unreacted H₂O₂. The
127 obtained OCG was subsequently freeze dried and stored until further use.

128 *2.3. Determination of aldehyde and carboxyl content*

129 The aldehyde group content of the prepared OCG was determined via UV absorption
130 method as previously described [33]. OCG was reacted with hydroxylamine hydrochloride
131 and the maximum UV absorbance can be determined at 233 nm using a UV-Vis single beam
132 spectrophotometer (Model I-290), attributed to the formation of π-π conjugation in the
133 molecular structure of converted OCG to oxime via a Schiff's base reaction. The calibration
134 curve was subsequently established using different volumes of 4.22 × 10⁻⁴ M glyoxal
135 solution. 2 mL of 1.5% (w/w) Ca(CH₃COO)₂ and 2 mL of 0.2% (w/w) hydroxylamine

136 hydrochloride were added into the different volumes of glyoxal solutions. The mixtures were
 137 reacted at 50 °C for 20 min and then cooled. The obtained mixtures were diluted with 50 mL
 138 distilled water and measured at 233 nm absorbance. 10 mg of OCG samples were dissolved
 139 in 10 mL of distilled water and the determination of aldehyde content was analysed following
 140 the procedure described above. The absorbance calibration curve was established and the
 141 aldehyde content calculated using the equation: $y = 7488.8x + 0.0065$ (where, x is the
 142 aldehyde content and y is the UV absorbance).

$$\%CHO = \frac{(y - 0.0065) \times 0.05 \times 29}{M \times 7488.8} \times 100 \quad (1)$$

143 Where, M is the dry weight (g) of OCG and 29 is the molecular weight of the aldehyde group.
 144 All tests were performed in triplicates and the average values recorded.

145 The determination of the carboxyl content was performed according to previous reported
 146 calcium-acetate consumption method with slight moderations [31, 34]. In essence, carboxyl
 147 groups can react with the salts of weak acids such as calcium acetate, forming a carboxyl salt
 148 and releasing a same amount of weak acid. Accordingly, 0.045 g of OCG was dissolved in 30
 149 mL of distilled water. 10 mL of prepared 0.1 M $\text{Ca}(\text{CH}_3\text{COO})_2$ was added and the mixture
 150 stirred for 1h to ensure complete reaction between OCG and $\text{Ca}(\text{CH}_3\text{COO})_2$. The mixture
 151 solution was then titrated against 0.01M NaOH from colourless to a stable pink colour using
 152 0.2 %w/v phenolphthalein as the indicator. A control sample analysis was conducted using
 153 non-oxidized CG. The percentage of carboxyl groups in the OCG sample was calculated
 154 following the equation below:

$$\%COOH = \frac{(V_{OX} - V_C) \times C_{NaOH} \times 45}{W} \times 100 \quad (2)$$

156 Where, V_{ox} and V_c are the volumes (L) of NaOH used for oxidized and control sample
 157 titrations, respectively. C_{NaOH} is the concentration (M) of prepared NaOH solution, W (g) is
 158 the weight of sample and 45 is the molecular weight of carboxyl group. Readings were
 159 performed in triplicates and the average value recorded.

160 2.4. Preparation of gels

161 The compositions of the different gels prepared are presented in Table 1. Firstly, 7 %w/v
 162 gelatin stock solution was prepared by complete dissolution in phosphate buffer saline (PBS)

163 pH 7.4 solution at 80 °C for 30 min. The solution was cooled to 60 °C and under gentle
 164 stirring, varying amounts of OCG was added. The mixture was continuously stirred and the pH
 165 adjusted in the range 5 to 5.5, in order to promote the reaction between aldehyde and amine
 166 functional groups. Subsequently, wet bacterial cellulose was added to the reaction mixture in
 167 different weight ratios. Finally, the homogenous mixtures were transferred into petri dishes
 168 and allowed to cool to room temperature forming gel samples without BC (GC1 and GC2)
 169 and samples interpenetrated with BC (GCB1, GCB2 and GCB3). The formed solid gel
 170 samples were then washed multiple times with distilled water to remove unreacted
 171 components and freeze dried to form gel scaffolds. The prepared gels and freeze-dried
 172 scaffolds were stored and used for further analyses.

173

174 **Table 1**

175 Compositions of the all-natural prepared gels.

Sample	Gelatin (%w/v)	OCG (%w/v)	BC (%w/w)
GC1	7	0.5	/
GC2	7	1	/
GCB1	7	1	0.5
GCB2	7	1	1
GCB3	7	1	2

176

177 *2.5. Gelation kinetics and swelling analysis*

178 In this study, gelation as a function of time was evaluated via the inverted tube test method
 179 as described previously with slight moderation [35, 36]. In brief, 1 mL of the different gel
 180 solutions prepared as described above were transferred into 1.5 mL vials. The vials were
 181 further incubated in a temperature-controlled bath for a period of time. The sol-gel transition
 182 time of the gel samples was determined by interval control of the test vials in the bath and
 183 inverted every minute. If the test vial containing the solution is tilted and a deformation flow
 184 occurs, is defined as a sol phase, while if no flow occurs it is described as a gel phase. The
 185 time at which the gel did not flow was recorded as the gelation time.

186 The equilibrium swelling capacities of the prepared gels was measured by immersing the
 187 freeze-dried weighted samples in PBS (pH 7.4) at 37 °C and at different time intervals of 6,
 188 12 and 24h, the samples were extracted, blotted with tissue paper to remove excess water and
 189 weighed. Readings were performed in triplicate and the average swollen value recorded. The
 190 equilibrium swelling percentages (ES%) were calculated using the equation below:

$$ES\% = \frac{W_t - W_i}{W_i} \times 100 \quad (3)$$

191 Where, W_t is the weight of the swollen gels after 24h and W_i is the initial weight of the
192 freeze-dried gels.

193 2.6. Viscoelastic and mechanical properties of gels

194 Rheological measurements were conducted on a rotational rheometer (Anton Paar MCR
195 502) with parallel plate geometry of 25 mm diameter. Initially, an amplitude sweep
196 measurement was conducted to determine the linear viscoelastic range (LVE) for the gels.
197 Next, storage modulus (G') and loss modulus (G'') were measured as a function of change in
198 temperature from 18 to 50 °C at a fixed heating/cooling rate of 2 °C/min and constant
199 frequency of 1 Hz in the LVE of 1% of strain. Finally, strain sweep studies were performed
200 by varying the amplitude from 0.01 to 100% at constant frequency of 1 Hz. Two replicate
201 measurements of each gel sample were performed and the average value of the obtained
202 results over time was determined. The compression modulus of elasticity of the gels was
203 evaluated by unconfined compression at 25 °C using a Testometric MT350-5CT
204 (Labomachine, Czech Republic). Prior to analysis, the formed solid gels were severally
205 washed with distilled water and cut into cylindrical shapes. The compressive modulus was
206 then measured under a static load of 5 kg and crosshead speed of 1 mm/min.

207 2.7. Characterization of gel scaffolds

208 Fourier transform infrared (FTIR) spectra of the pristine polymers and gel scaffold
209 samples were recorded using a Nicolet iS5 (Thermo Scientific, USA), scanned at a resolution
210 of 4.0 cm^{-1} in the range of 4000–400 cm^{-1} . The surface morphology of the gel scaffolds was
211 observed by scanning electron microscopy (SEM) using a bench-top PhenomTM Pro
212 microscope operating at 10 kV. The crystallinity of the samples was analysed by a high-
213 resolution Mini FlexTM 600 X-ray diffractometer (Rigaku, Japan). The scans were conducted
214 in the range of 0 - 90 °, at a speed of 5 °/min using a Cobalt radiation at 40 kV and 15 mA.
215 Thermogravimetric analysis (TGA) was conducted using a Q500 thermogravimetric analyser
216 at a heating rate of 10 °C/min from 25 to 600 °C under a nitrogen flow rate of 40 - 60
217 mL/min.

218 2.8. Drug loading efficiency, drug release and kinetics study

219 BSA was used as the model protein for drug delivery investigation. The initial loading
220 concentration of BSA in the gels was 5 mg/mL. Briefly, BSA was incorporated into the gels
221 by first heating the gel mixture to 60 °C, followed by cooling to ≈ 40 °C, and then loading of

222 BSA into the gel systems added under gentle stirring. The gel solutions were allowed to stir
 223 for 2h and further cooled down to room temperature to obtain solid BSA-loaded gels. The
 224 loaded samples were then freeze dried and the obtained scaffolds used for further analysis. In
 225 order to calculate the drug loading efficiency, the BSA-loaded gel scaffolds were weighed,
 226 immersed in PBS and crushed to release the drug. The mixture was centrifuged at 14000 rpm,
 227 the supernatant collected and the absorbance determined at 280 nm (corresponding to the
 228 absorbance of tyrosine and tryptophan) using a UV-VIS single beam spectrophotometer
 229 (Model I-290). Using a standard calibration curve of BSA in PBS, the drug loading efficiency
 230 percentage (DLE%) was calculated according to the equation below:

$$\text{DLE}\% = \frac{\text{Total amount of BSA added} - \text{Amount of BSA in supernatant}}{\text{Total amount of BSA added}} \times 100 \quad (4)$$

231
 232 The *in vitro* release study was conducted according to previous reported procedures with
 233 slight modification [37, 38]. In brief, BSA release from the prepared hydrogels was studied in
 234 PBS of pH 7.4 using a shaking water bath (100 rpm) at 37 °C. BSA-loaded hydrogels were
 235 immersed in 100 mL of PBS solution and placed in the water bath for release. At
 236 predetermined time intervals, 1 mL of release medium was collected and the amount of BSA
 237 analysed spectrophotometrically using a UV-VIS single beam spectrophotometer (Model I-
 238 290) at 280 nm. In order to maintain a constant release medium volume, the collected sample
 239 for analysis was replace with same amount of fresh PBS solution. The amounts of released
 240 BSA were determined from the calibration curve and the cumulative drug release percentage
 241 (CD%) was calculated according to the equation below:

$$\text{CD}\% = \frac{D_t}{D_0} \times 100 \quad (5)$$

242 Where, D_t is the total amount of BSA released at time t and D_0 is the initial amount of BSA in
 243 the gel scaffolds. In order to establish the mechanism of drug release kinetics, the release data
 244 of BSA from the gel scaffolds were computed and fitted to four widely use kinetic models;
 245 Zero-order (Eq. 6), First-order (Eq. 7), Higuchi (Eq. 8) and Korsmeyer-Peppas model (Eq. 9)
 246 [39-41]. The mathematical expression is provided below:

$$M_t = K_0 t \quad (6)$$

$$\ln M_t = \ln M_0 - K_1 t \quad (7)$$

$$M_t = K_H t^{1/2} \quad (8)$$

$$\ln M_t = n \ln t + \ln K_{KP} \quad (9)$$

247 Where, M_t is the amount of drug released at time t , M_o is the initial concentration of drug and
 248 the parameters K_o , K_I , K_H and K_{KP} are the rate constants related to the cumulative drug
 249 release for Zero-order, First-order, Higuchi, and Korsmeyer-Peppas models. n is an indicator
 250 of the drug release mechanism, that is a value of $n \leq 0.5$ indicates that drug release is
 251 controlled by Fickian diffusion. A value of $n \geq 1.0$ suggests a non-Fickian diffusion with
 252 characteristic of zero-order release rate and values in the range $0.5 < n < 1.0$, indicates that
 253 the release process is anomalous. Values of $n = 0.5$ and 1.0 describes case I and II transport
 254 processes, respectively, which are characterized by polymer relaxation related to polymer
 255 erosion during enzymatic degradation [42, 43]. Considering Korsmeyer-Peppas model is
 256 valid only within the 60% range of the cumulative drug release curve, all studied kinetic
 257 models were fitted to the first 60% cumulative release data of the drug release curve [44].

258 2.9. *In vitro* cell cytotoxicity, enzymatic degradation and antibacterial analysis of gel 259 scaffolds

260 To evaluate the cytocompatibility of the prepared gel scaffolds, the samples were cut into
 261 15 mm diameter disks, swollen in culture medium for 2h and the extracts prepared (0.1
 262 mg/mL of culture medium) according to ISO standard 10993-12. ATCC-formulated
 263 Dulbecco's Modified Eagle's (DMEM) medium (Biosera, France) containing 10% of calf
 264 serum and 100 U/mL penicillin/streptomycin (Biosera, France), was used as the culture
 265 medium. The extract was diluted with the culture medium to obtain following concentration:
 266 50, 75 and 100 % of parent extract. All assays were performed in triplicates with the extracts
 267 used within 24h. Mouse embryonic fibroblast cells (ECACC 93061524, England) were
 268 seeded to pre-incubated microtitration test plate dishes (TPP, Switzerland) at a concentration
 269 of 1×10^5 cells/mL. The sucked-up culture medium was replenished. The cell viability was
 270 measured using Tetrazolium (MTT cell proliferation assay kit, Duchefa Biochemie,
 271 Netherlands). The absorbance of the solutions was measured spectrophotometrically at 570
 272 nm. The results are presented as reduction of cell viability in relative values when compared
 273 to cells cultivated in medium without the extracts of tested samples.

274 The degradation of the gel scaffolds was investigated with respect to weight loss in PBS
 275 pH 7.4 aqueous solution over a period of 14 days. Initially, 0.1 % w/v enzymatic solutions

276 were prepared by dissolving calculated amounts of lysozyme in PBS. The freeze-dried gel
277 scaffolds were weighed (W_o), immersed in the PBS solutions and incubated under mild
278 shaking of 100 rpm in an oven at 37 °C. At specified time intervals, the gel samples were
279 removed from the PBS solution, gently blotted with filter paper, dried and weighed (W_t). In
280 order to simulate continuous lysozyme activity, 5 mL of the degradation solution was
281 refreshed every 2 days. Readings were performed in triplicates and the average values
282 recorded. The extent of in vitro degradation was expressed as the percentage weight loss
283 (%WL) using the equation $((W_o - W_t)/W_o) \times 100$.

284 Antibacterial activity of prepared non-loaded and loaded gel scaffolds was assessed
285 against *Escherichia coli* (CCM 4517), *Staphylococcus aureus* (CCM 4516), and *Klebsiella*
286 *pneumonia* (CCM 4415) using the agar disc diffusion test. Typical, 100 μ L aliquot of each
287 bacterial stock suspension of approximate concentration 12×10^8 cells/mL was uniformly
288 spread on a tryptone soya agar plate and the gel scaffold samples (6 mm diameter) placed on
289 top of the plates. The plates were then incubated at 37 °C for 18h and the inhibition zones
290 measured.

291 2.10. Statistical analysis

292 All experimental data were analysed using Analysis of Variance (ANOVA). Statistical
293 significance was evaluated at p-value ≤ 0.05 . All results are presented as mean \pm standard
294 deviation.

295 3. Results and discussion

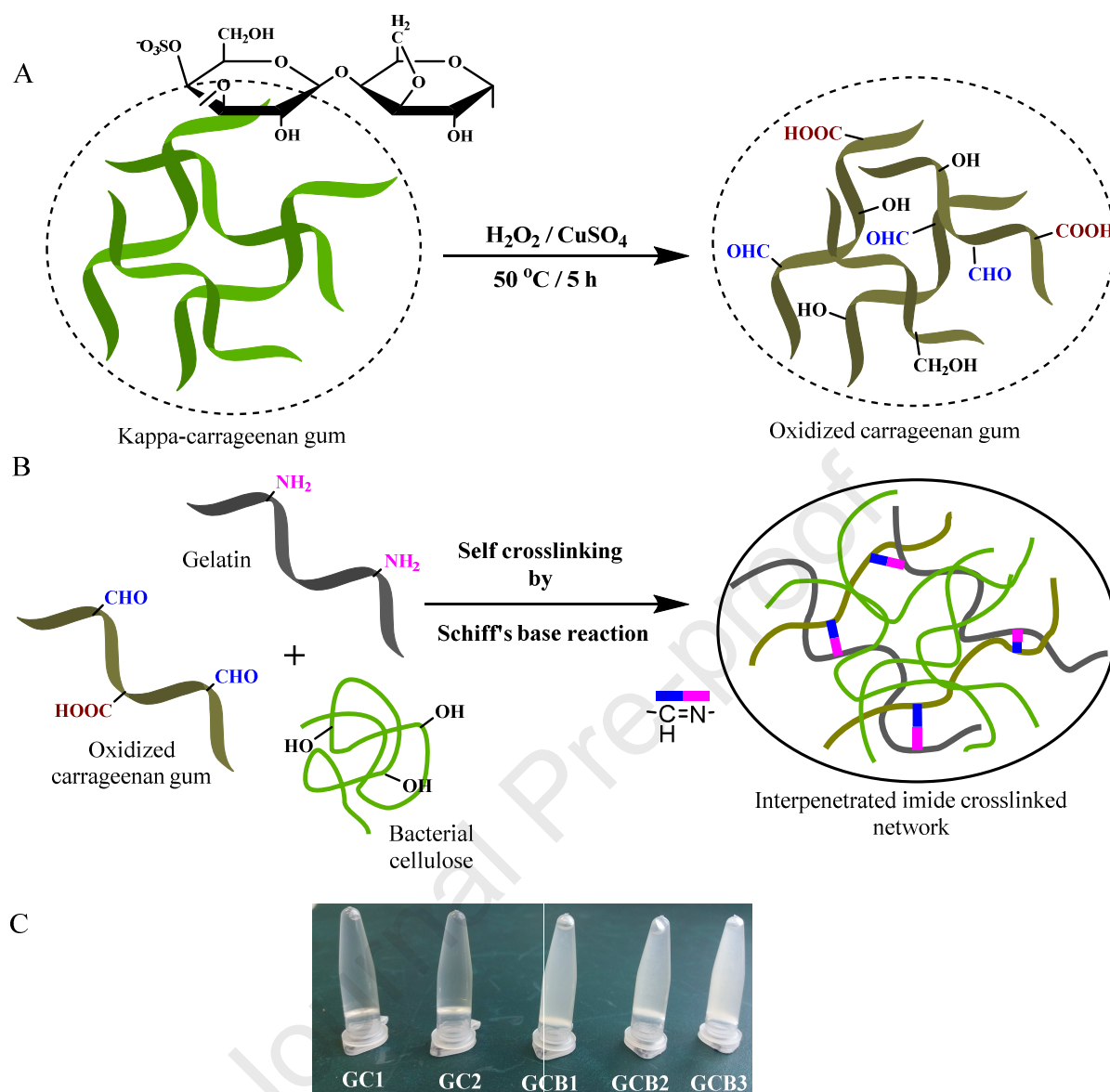
296 3.1. Gel fabrication and gelation kinetics

297 Based on the synthesis of OCG from CG in the present study, a possible chemical reaction
298 phase is presented in Fig. 1. In essence, the oxidation of CG was performed using H_2O_2 and
299 $CuSO_4$ as the oxidant and catalyst, respectively [31, 45]. The reaction mechanism occurred
300 via a radical initialized process where in the presence of copper ions, H_2O_2 quickly
301 decomposed to different radicals such as $HO\cdot$, $HO_2\cdot$, and $O_2\cdot$. These generated reactive free
302 radicals rapidly reacted with the hydroxyl groups of CG, resulting in the formation of
303 aldehyde (CHO) and carboxyl (COOH) groups on the backbone chain of CG [46]. The
304 content percentages of CHO and COOH for the final oxidized product was determined as
305 1.04 ± 0.06 and $11.60 \pm 0.03\%$, respectively. This indicated that CHO and COOH groups
306 were successfully incorporated into CG structure. Subsequently, the preparation of the

307 injectable gels was carried out by crosslinking gelatin and OCG followed by fibrillar
308 interpenetration with bacterial cellulose (Fig. 1a). The crosslinking process predominantly
309 occurred due to Schiff's base reaction between the free ϵ -amines ($-\text{NH}_2$) of lysine or
310 hydroxylysine side groups of gelatin and the grafted aldehyde groups of OCG forming imide
311 bonds [47]. By physical visualization, it was observed that the viscosity of the crosslinked
312 mixture between gelatin and OCG gradually increased with time forming weak translucent
313 gels. These weak formed gels were further improved by fibrillar interpenetration with
314 bacterial cellulose (Fig. 1b), which in turn increased the viscosity of the mixture and formed
315 gels with enhanced mechanical stability that could self-support its own weight at 37 °C.

316 Gelation rates of the different prepared gels were monitored at 37 °C and the time of gel
317 formation determined as 23, 15, 12, 7 and 4 minutes for investigated GC1, GC2, GCB1,
318 GCB2 and GCB3, respectively. As observed, the gelation time of GC gels were longer
319 compared to that of GCB. Gelation time of GC gels occurred between 15 to 25 min as
320 compared to GCBs that was in the range of 4 to 12 min. The fast gelation in GCB gels was
321 attributed to the incorporation of BC. The effect of BC was confirmed as the gelation time
322 gradually decreased with increase in BC ratio. As such, GCB3 with the highest BC content
323 reached complete gelation within 4 min, exhibiting a considerably faster gelation rate than the
324 other gels. It is obvious that the addition of BC in the crosslinked gel systems significantly
325 reduced the gelation time. This fast gelation may be attributed to the increase in total
326 viscosity of the gel mixtures that in turn enhanced intra/intermolecular hydrogen bond
327 interaction, forming a high dense 3D physical and chemical crosslinked network. The images
328 of the formed gels after their gelation times are displayed in Fig. 1c.

329



330

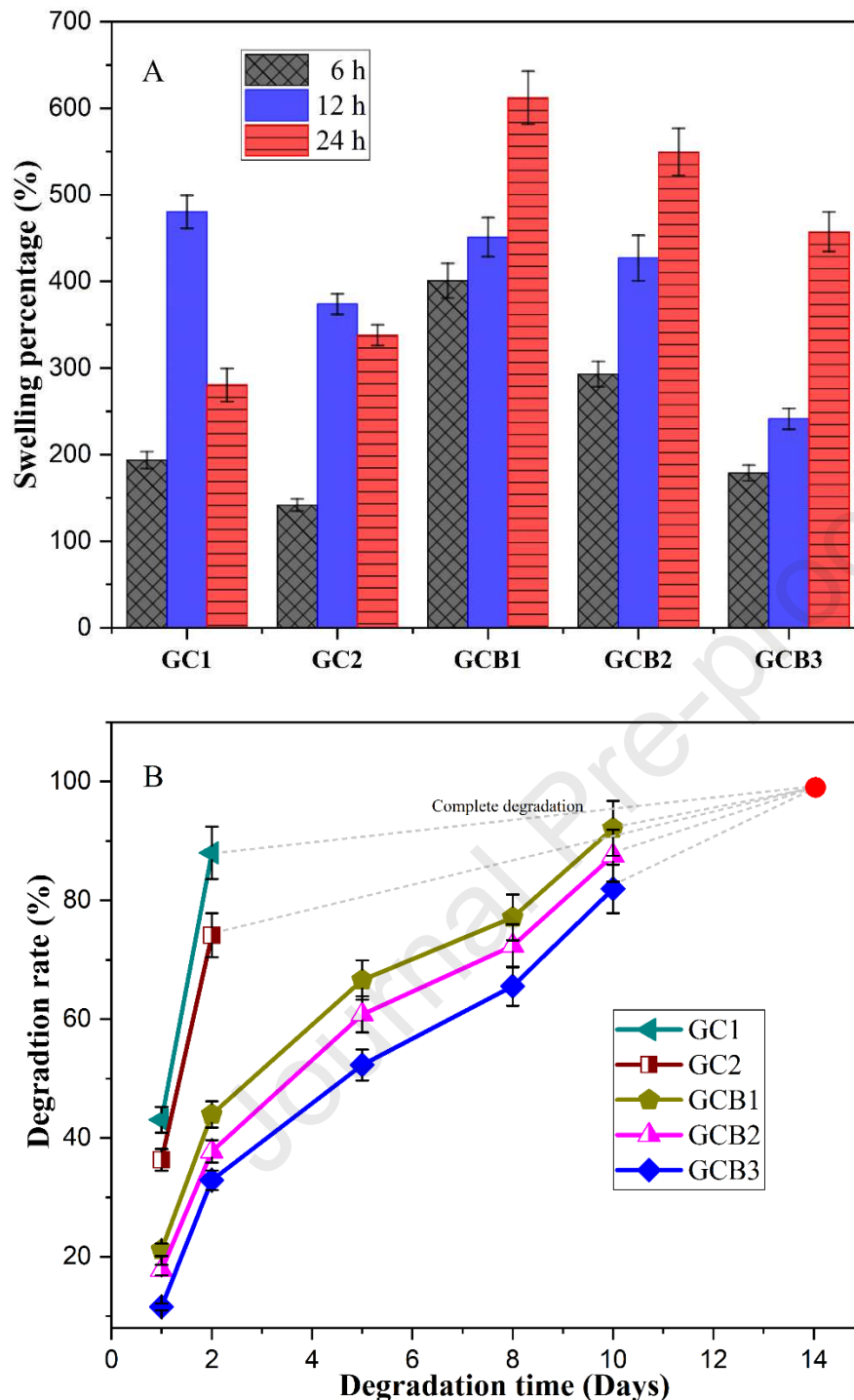
331 **Fig. 1.** a) Schematic representation for the synthesis of oxidized carrageenan gum, b)
 332 proposed crosslinked network in gels via Schiff base reaction and c) the images of the
 333 different formed gels at 37 °C.

334 3.2. *In vitro* swelling and degradation analysis of freeze-dried gel scaffolds

335 Water absorption ability is an important parameter to predict the movement of oxygen and
 336 nutrients within a given crosslinked gel structure [48]. Fig. 2a shows the swelling results of
 337 the prepared freeze-dried gel scaffolds swollen in dH₂O as a function of time to reach
 338 equilibrium water uptake. According to the results obtained, maximum swelling was reach
 339 within 12 h for GC gel scaffolds. However, GC1 showed higher swelling when to GC2,
 340 which can be attributed to the low crosslink density of GC1. After 24 h, GC gels depicted
 341 drastic decrease in swelling capacity and such possible occurrence could be attributed to the
 342 gradual dissolution/degradation of the gel structural network considering gelatin is

343 temperature sensitive. Notwithstanding, an increase in the crosslink density in GC2 showed
344 slower dissolution as compared to GC1. Following the incorporation of BC in the gel
345 systems, higher swelling values were achieved after 24 h. But a decrease in water uptake was
346 observed within the GCB gel scaffolds as BC content increased. The possible reason
347 associated to this relates to the formation of more hydrogen bonds as well as the effect of
348 inter/intramolecular interaction in the gel network as BC amount increased, which in turn
349 restricted the swelling of the gel scaffolds. Overall, the maximum equilibrium swelling values
350 after 24 h were determined as 280.42 ± 19.02 , 340.98 ± 11.89 , 620.14 ± 30.60 , $550.48 \pm$
351 27.47 and $460.38 \pm 22.86\%$ for GC1, GC2, GCB1, GCB2 and GCB3, respectively.

352 *In vitro* biodegradation of the gels and their scaffolds is considered a crucial parameter
353 particularly in wound healing applications. This is because the gel provides mechanical
354 strength support to the wound, protects the wound against infections and aids in the release of
355 loaded bioactive molecules that promote wound healing [28]. In the present study, enzymatic
356 degradation studies were performed to determine the stability of the prepared gel scaffolds.
357 Herein, lysozyme was employed as the degradation enzyme prepared at a concentration of
358 0.1 %w/v to mimic an *in vivo* degradation system and the weight loss of the gel scaffolds
359 were monitored in PBS at 37 °C within a period of 14 days. Fig. 2b depicts the degradation of
360 GC and GCB gel scaffolds based on weight loss as a function of time. As expected, a faster
361 weight loss was observed for gel scaffolds without BC (GC1 and GC2) compared to samples
362 with BC (GCB1, GCB2 and GCB3). According to observed results, GC1 and GC2
363 completely degraded in less than 5 days while GCB gel scaffolds showed more than 50%
364 weight loss on the 5th day. On the 5th day of incubation, the weight loss of GCB gel scaffolds
365 was determined as $66.58 \pm 3.32\%$, $60.80 \pm 3.04\%$ and $52.29 \pm 2.61\%$ for GCB1,
366 GCB2 and GCB3, respectively.



367

368 **Fig. 2.** a) Swelling capacity and b) enzymatic degradation of prepared gel scaffolds as a
 369 function of time.

370 Thus, it is evident that the incorporation of BC increased the stability of GCB gel scaffolds,
 371 which in tend decreased degradation rate as compared to GC samples. This gel matrix
 372 stability may be attributed to induced fibrillar meshwork structure formation and
 373 intra/intermolecular hydrogen bonding introduced by BC that retarded degradation of the
 374 GCBs. This contributes to the beneficial properties of this material for application during

375 wound healing application. Overall, the first 5 days of degradation were mostly attributed to
376 sufficient swelling of the gels in solution and thereafter, the degradation increased due to
377 gradual disintegration of gel structural network via crosslinked sites. GCB gel scaffolds
378 showed >80% weight loss after 10 days with complete degradation within 14 days. Based on
379 the swelling and degradation studies discussed, GC2 (as control) and GCB2 samples were
380 further analysed in the present study and subsequently designated as GC and GCB,
381 respectively.

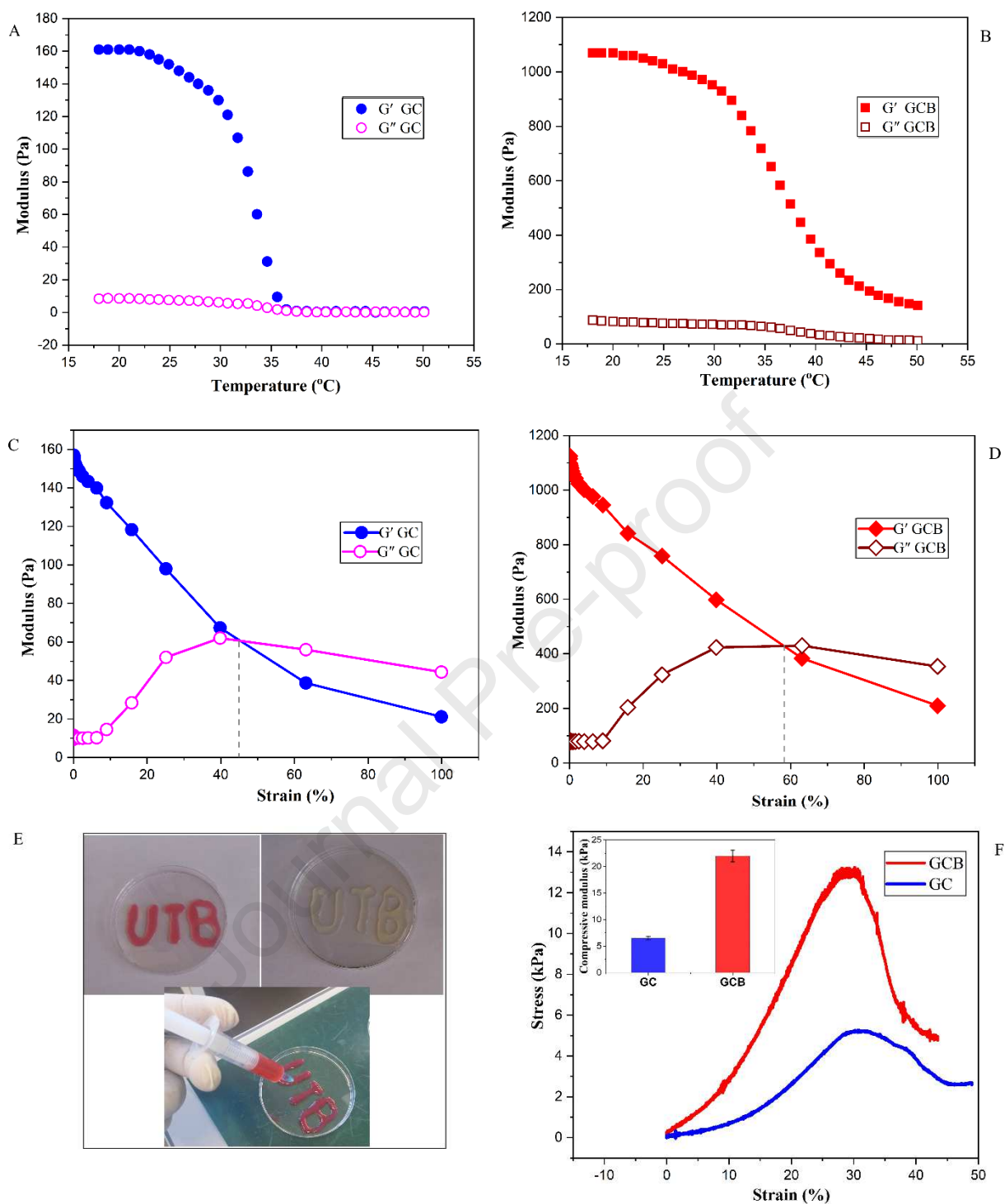
382 3.3. Characterization techniques

383 3.3.1. Rheological and compression properties of prepared gels

384 The rheological properties of the prepared gels were analysed via oscillatory rheology to
385 demonstrate the stability of the samples after shear deformation. Fig. 3a and b displays the
386 temperature sweep tests of storage modulus G' and loss modulus G'' within a temperature
387 range of 10 to 50 °C. According to literature [1], gelatin is unable to form gels at body
388 temperature ($G' < G''$) but forms gels at lower temperature below 27 °C ($G' > G''$). However,
389 by crosslinking gelatin with a small amount of carrageenan gum causes a big difference
390 whereby the obtained gels (GC) show resistance to temperature evolution where G' is larger
391 than G'' in the range of 10 – 35 °C, indicating the effect of oxidized carrageenan gum in the
392 gelling process of gelatin. By adding BC to the gel matrix to produce GCB, higher G' values
393 are achieved compared to that of GC gels. This increase in G' may be attributed to
394 intermolecular complex formation between the chemical crosslinked network and BC via
395 physical crosslinking. At temperature ≥ 37 °C, G' of GCB gels was steadily larger than G'' ,
396 suggesting the network of the gels are less thermosensitive as compared to GC that are less
397 stable. In addition, amplitude sweep studies were performed by subjecting the gels to varying
398 strains ranging from 0.1% to 100%. In conformity to Fig. 3c and d, these gels showed signs
399 of crossover at around strain values of 45 and 58% for GC and GCB gels, respectively.
400 Within 100% strain, the addition of BC led to a higher elasticity, which probably equipped
401 the GCB gels with the more capability to withstand administration-related strains as an
402 injectable formulation. Moreover, the rigidity in chemical structure of some natural
403 polysaccharides such as carrageenan gum, enables them to be use utilized as modifiers for
404 tuning rheological properties thereby endowing the prepared gels with desirable shear-
405 thinning properties [49]. In addition, bacterial cellulose greatly contributed to the shear
406 thinning properties via its fibrillar structures that can re-order in the flow direction of the gels

407 as they display decrease in viscosity with loss in their network structure [50]. As such, the
408 prepared GCB gels in the present study can be extruded via injection needles forming
409 different patterns (Fig. 3e). The extruded gels quickly reassemble within minutes upon
410 cessation of shear. This result is in accordance with the rheological test and may be attributed
411 to the good shear-thinning properties arising from carrageenan gum and bacterial cellulose
412 that have been widely applied as injectable matrix for tissue regeneration in wound healing
413 application [29, 51].

414 The mechanical properties of wet GC and GCB hydrogels were determined under
415 compression at 80% strain. Fig. 3f displays the results obtained during the compression
416 process of the gels. From the stress–strain curves and compressive modulus, the incorporation
417 of BC in the crosslinked gel matrix showed great effect on the stress–strain behaviour and
418 compressive modulus of elasticity of the prepared gels. Based on deduced results, the
419 compressive modulus was determined as 6.5 ± 0.35 and 22.3 ± 1.40 kPa for GC and GCB
420 gels, respectively. This indicated that the mechanical properties of the gels were significantly
421 improved due to interpenetration of GC gel matrix with BC [52]. Overall, the addition of BC
422 in the gels greatly enhanced the mechanical stability of the gels making it suitable for
423 application as injectable formulation.

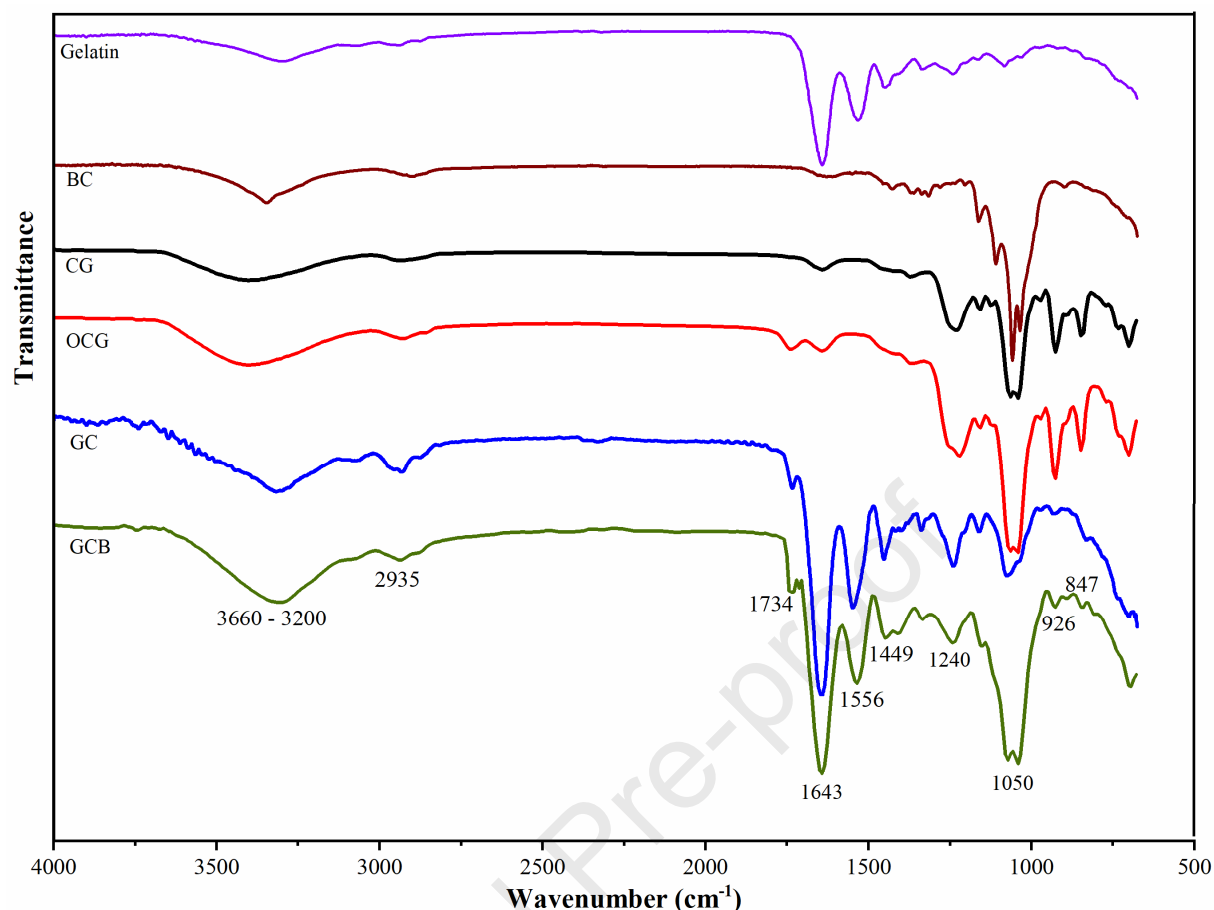


424

425 **Fig. 3.** Rheological measurements of GC (control) and GCB gels a) and b) temperature sweep
 426 at constant frequency of 1.0 Hz and strain of 0.1%, c) and d) strain sweep at constant
 427 frequency of 1.0 Hz and temperature 25 °C, e) images of dyed and non-dyed GCB gel
 428 extruded from a 22-gauge syringe forming different patterns and f) the stress-strain curves
 429 and compressive modulus of elasticity of GC and GCB gels at 25 °C.

430 3.3.2. Analysis of freeze-dried gel scaffolds

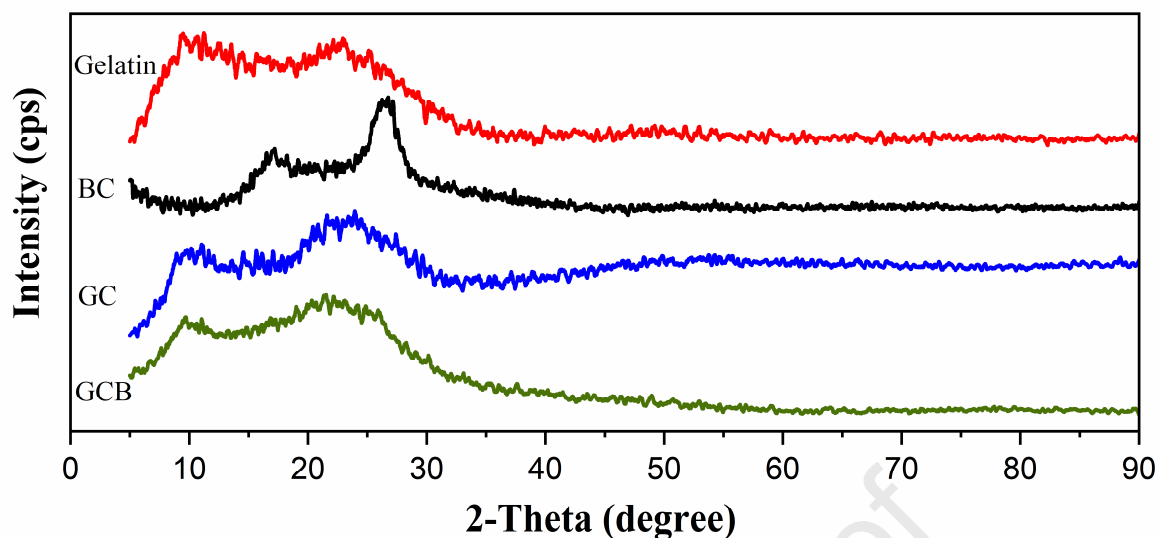
431 The FTIR spectra of the different samples are presented in Fig. 4. According to obtained
432 results, CG and OCG displayed characteristic peaks at around 3426, 2935, 1240, 928 and 840
433 cm^{-1} . The strong broad band at around 3426 cm^{-1} is attributed to hydroxyl (OH) group
434 stretching. The peak at 2935 cm^{-1} is assigned to the asymmetric stretching vibration of
435 methyl and methylene groups. The peaks at 1240 and 847 cm^{-1} are assigned to the symmetric
436 vibration of $-\text{SO}_4$ and $\text{C}_4-\text{O}-\text{S}$ attached on β -D-galactose carrageenan structure, respectively
437 [31]. The peak at 1050 and 928 cm^{-1} relates to the asymmetric stretching vibration of C–O
438 and C–O–C of 3,6-anhydro-D-galactose in CG. By comparing CG and OCG, results suggest
439 that the primary structure of carrageenan gum was not destroyed during oxidation. However,
440 a new characteristic peak appeared around 1734 cm^{-1} in the spectrum of OCG, which is
441 attributed to the stretching vibrations of C=O [53]. This confirms that carboxyl and aldehyde
442 groups were successfully introduced in the backbone structure of CG via selective oxidation.
443 By further analyses of GC and GCB spectra, characteristic peaks of pristine gelatin were
444 observed. This includes the peak at 1734 cm^{-1} that relates to C=O stretching of β -lactam,
445 1643 cm^{-1} ascribed to both C–O stretching of primary amide and C–C stretching of benzene
446 ring and the peak at 1556 cm^{-1} that describes both N-H in-plane bending and C=N stretching
447 of secondary amide groups in the structure of gelatin [54]. While the broad peak between
448 3500 and 3200 cm^{-1} corresponds to the overlapped stretching vibrations of N-H from gelatin
449 and OH groups of OCG and BC. In addition, the peak at 1244 and 1050 cm^{-1} are attributed to
450 the C–O asymmetric bridge stretching and C–O–C vibration of OCG and BC. However, the
451 stretching vibrations of the imine group (C=N) formed via Schiff base reaction from the
452 reaction of primary amino groups of gelatin and the aldehyde groups of OCG usually appears
453 around between 1640 to 1520 cm^{-1} , but was overlapped by the strong functional C=N group
454 peaks of pristine gelatin in this region [55].



455

456 **Fig. 4.** FTIR spectra of investigated samples.

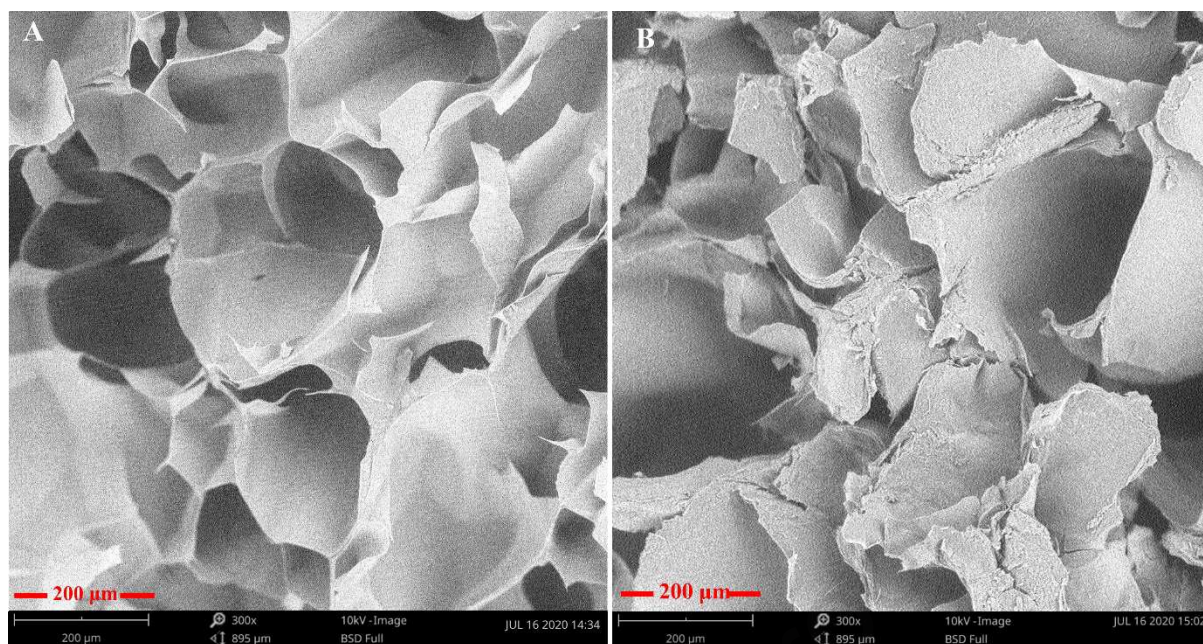
457 Fig. 5 displays the X-ray diffractograms Gelatin, BC, GC and GCB. According to
 458 literature [56], a polymer matrix with sharper and stronger peaks depicts higher degree of
 459 crystallinity, while polymeric systems with relatively weak and wide range peaks shows the
 460 presence of amorphous regions. As observed, the diffractograms of the prepared GC and
 461 GCB gel scaffolds were mainly dominated by two peaks at $2\theta = 9.60^\circ$ and 21.39° relating to
 462 the partial crystallinity characteristic of gelatin. These characteristic peaks are generally
 463 assigned to the triple-helical crystalline structure in gelatin [57, 58]. Following the results
 464 obtained, GC displayed lower peak intensity of ≈ 480 counts at $2\theta = 21.39^\circ$ compared to ≈ 675
 465 counts determined for GCB. This additionally confirms the effect of incorporated BC in the
 466 gel structure that in tend increased the degree of crystallinity GCB matrix. Furthermore, the
 467 shift of the peaks corresponding to BC in GCB confirms the homogeneous interaction
 468 between BC fibrillar mesh network with the crosslinked gel matrix [59].



469

470 **Fig. 5.** X-ray diffractograms of gelatin, BC and prepared gel scaffolds (GC and GCB).

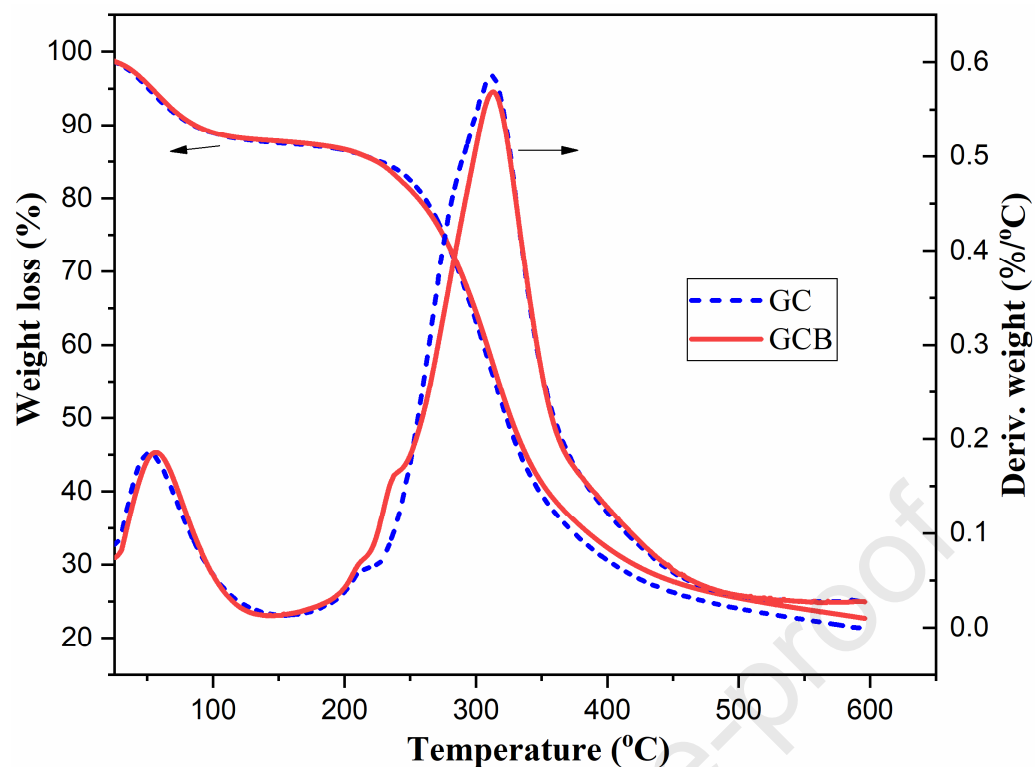
471 The surface morphologies of GC and GCB gel scaffolds were examined and the SEM
472 images are shown in Fig. 6a and b. Both gel scaffolds showed fairly regular, aligned and
473 straight channels with continuous micrometer size honeycomb-like morphology. In
474 conformity, it can be observed that the cross-sectional morphology GC was slightly different
475 from that of GCB. According to the figures, a denser and fibrillar mesh network was
476 displayed by GCB gel scaffolds. This can be attributed to the homogenous interpenetration
477 BC fibrils within the crosslinked GC matrix. Similar results on the preparation microbial
478 cellulose blended gelatin materials has been previously reported by Taokaew et al. [60]. In
479 addition, the prepared GC and GCB gel scaffolds demonstrated to be porous with average
480 pore diameters of 200 μm , which tissue regeneration may favour cell migration and
481 differentiation within the gel scaffolds.



482

483 **Fig. 6.** SEM images of freeze-dried a) GC and b) GCB gel scaffolds.

484 The thermal behaviour of GC and GCB gel scaffolds is of importance with regards to their
485 stability and controlled drug release mechanism. According to Treesuppharat et al. [61], the
486 thermal behaviour of hydrogels depends majorly on the thermo-responsive capability of their
487 crosslinked network. The TG and DTA curves of GC and GCB gel scaffolds are displayed in
488 Fig. 7. As observed, the variation in weight loss can be classified into three different phases.
489 From 25 to 200 °C, the weight loss was related to evaporation of free water molecules within
490 the gel structure. From 200 to 500 °C, the sharp weight loss was attributed to polymeric chain
491 and crosslinked network degradation. At temperature above 480 °C, relates to the thermal
492 decomposition associated with polymeric backbone leading to the formation of char residue.
493 Thus, it is suggested that the usage of GC and GCB gels for drug-delivery systems is suitable
494 at temperatures below 200 °C.



495

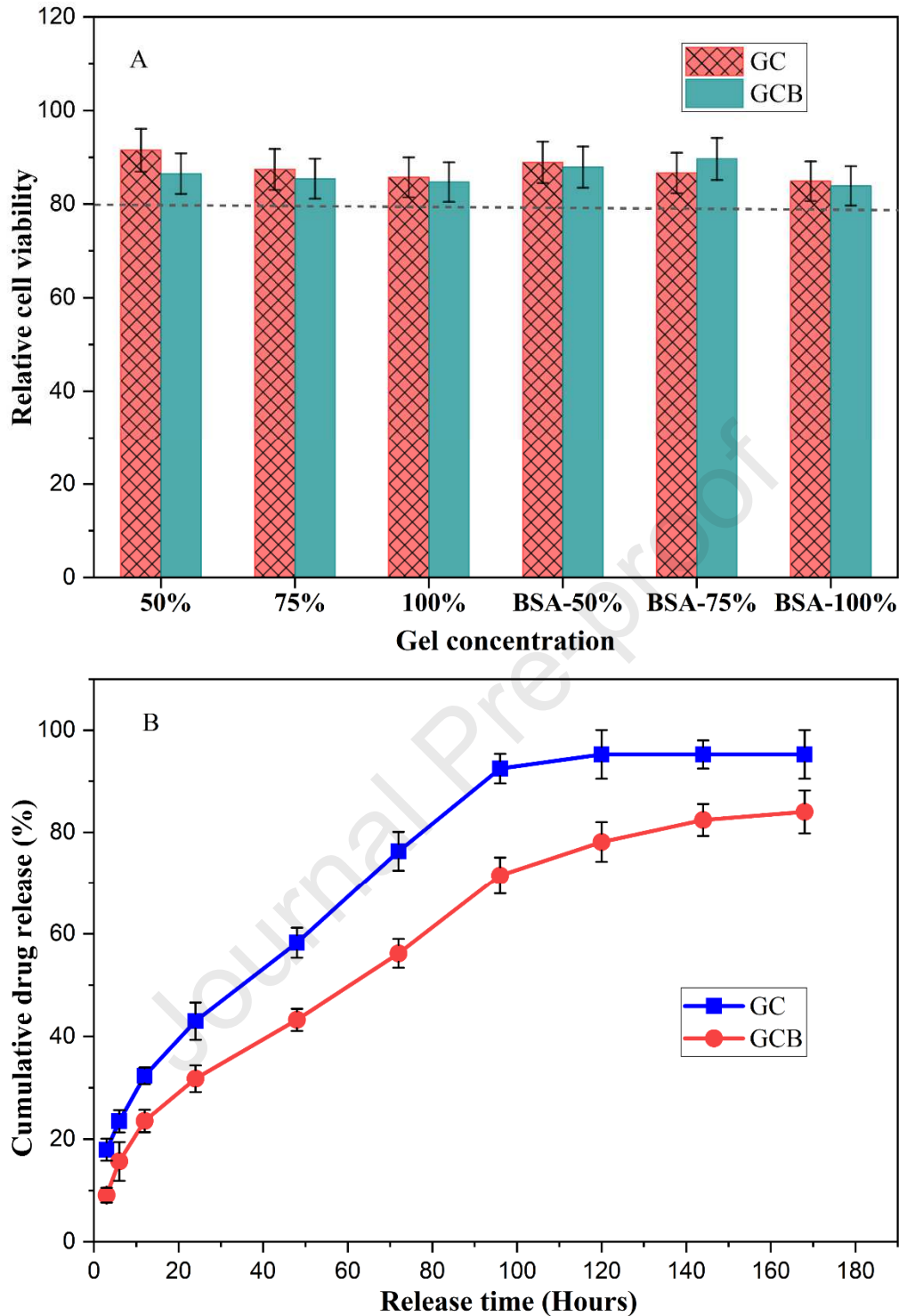
496 **Fig. 7.** Thermal degradation behaviour of GC and GCB gel scaffolds.

497 3.4. In vitro cell cytocompatibility and protein release study of gel scaffolds

498 Considering the prepared materials are applied as drug delivery vehicles and dressings,
 499 the toxicity of gels is investigated. In order to evaluate the cytotoxicity, different extract
 500 concentrations (50, 75 and 100%) of GCB (as control) and drug loaded gel scaffolds (BSA-
 501 GCB) were seeded with mouse embryonic fibroblast cells and measured by MTT assay.
 502 Based on deduce results, cell viability on GCB and BSA-GCB gel scaffolds depicted to be
 503 higher than 80% as shown in Fig. 8a. This indicated that GCB scaffolds are cytocompatible
 504 with low toxicity. Thus, this confirms the high efficiency of the prepared gel scaffold to
 505 promote cell growth and proliferation during tissue regeneration application. In addition, this
 506 revealed that the gel scaffold is suitable for maintaining proliferation of cells due to its
 507 similar physical structure and chemical composition as that of tissue matrix.

508 According to Kiortsis et al. [62], controlled release systems consist of a matrix-drug and
 509 polymeric assembly that follows a three-step release process; hydration of the drug matrix by
 510 release medium, swelling of the polymer matrix leading to disintegration, and lastly the
 511 transportation of the dissolved drug for release into the surrounding medium. In the present
 512 study, the drug delivery study was evaluated based on the release of BSA from GC (control)
 513 and GCB gel scaffolds. According to performed calculations, the drug loading efficiency of

514 BSA in GC and GCB gel scaffolds was determined as $69.2 \pm 2.43\%$ and $64.8 \pm 1.88\%$,
515 respectively. Fig. 8b shows the cumulative release of BSA from GC and GCB. As observed,
516 the initial burst release from GC ($17.37 \pm 1.64\%$) was approximately twice the release of
517 GCB ($9.33 \pm 2.01\%$) gel scaffold. This indicated that a sustain release process was achieved
518 for GCB by the incorporation of BC in the gel matrix. Based on the investigated release time
519 interval, cumulative drug release of $43.85 \pm 3.24\%$ and $31.53 \pm 1.77\%$ were reached within
520 24 h for GC and GCB, respectively. The low release rate from GCB may be attributed to
521 denser gel matrix and enhanced structural network crosslinking via intra/intermolecular
522 hydrogen bonding related to the incorporation of fibrillar BC. By comparing the release rates
523 of prepared gel scaffolds, GC showed increasing rapid release reaching more than 90%
524 cumulative release after 96 h. This rapid release process was attributed to the high
525 hydrophilicity of the gel causing the matrix to significantly swell and eventually disintegrate
526 [39]. On the other hand, GCB gel scaffold approached equilibrium released after 72 h with
527 maximum release determined as $84.01 \pm 3.66\%$ after 168 h. This indicates that the
528 results correlate with the mechanical stability and degradation capacity of the gel in relation
529 to interpenetration with BC. Overall, evaluations performed confirms that sustain release was
530 achieved by incorporating BC in the crosslinked GC gel matrix.



531
 532 **Fig. 8.** a) *In vitro* cell cytotoxicity evaluation based on viability of Mouse embryonic
 533 fibroblast cells grown on non-loaded GCB (as control) and drug-loaded GCB gel scaffolds at
 534 different concentrations in culture medium. Cell seeding concentration at 1×10^5 cells/mL. b)
 535 Cumulative release of BSA from GC and GCB scaffolds at 37 °C in PBS pH 7.4.

536 The mechanism of BSA release from GC and GCB gel scaffolds was studied by fitting the
 537 cumulative drug release data to four different release kinetic equations. The obtained results
 538 are displayed in Table 2. By comparing the calculated correlation coefficients (R^2) for

539 the different release kinetic models, it was observable that in both release medium the release
 540 kinetics best fitted with Higuchi's model ($R^2 = 0.999$ for GC and 0.998 for GCB). This
 541 reveals that the release of BSA from GC and GCB matrix based on the square root of time
 542 followed a Fickian diffusion process. Further, this phenomenon was supported by
 543 Korsmeyer-Peppas model ($R^2 = 0.999$ for GC and 0.991 for GCB). The n values of GC and
 544 GCB were determined as 0.427 and 0.486, respectively, which were <0.5 confirming the
 545 release mechanism of BSA predominantly followed Fickian's diffusion [63]. Typically, the
 546 release mechanism of Fickian diffusion describes that a substance in the matrix of a porous
 547 polymeric material can be released into immersion solution by diffusion via porous channels
 548 [64]. Considering BSA is water-insoluble, the hydration of GC and GCB via swelling opened
 549 the pore channels of the gel scaffolds thereby facilitating the diffusion of the drug into the
 550 release medium.

551 **Table 2**

552 Kinetic assessment of BSA release data from GC and GCB gel scaffolds.

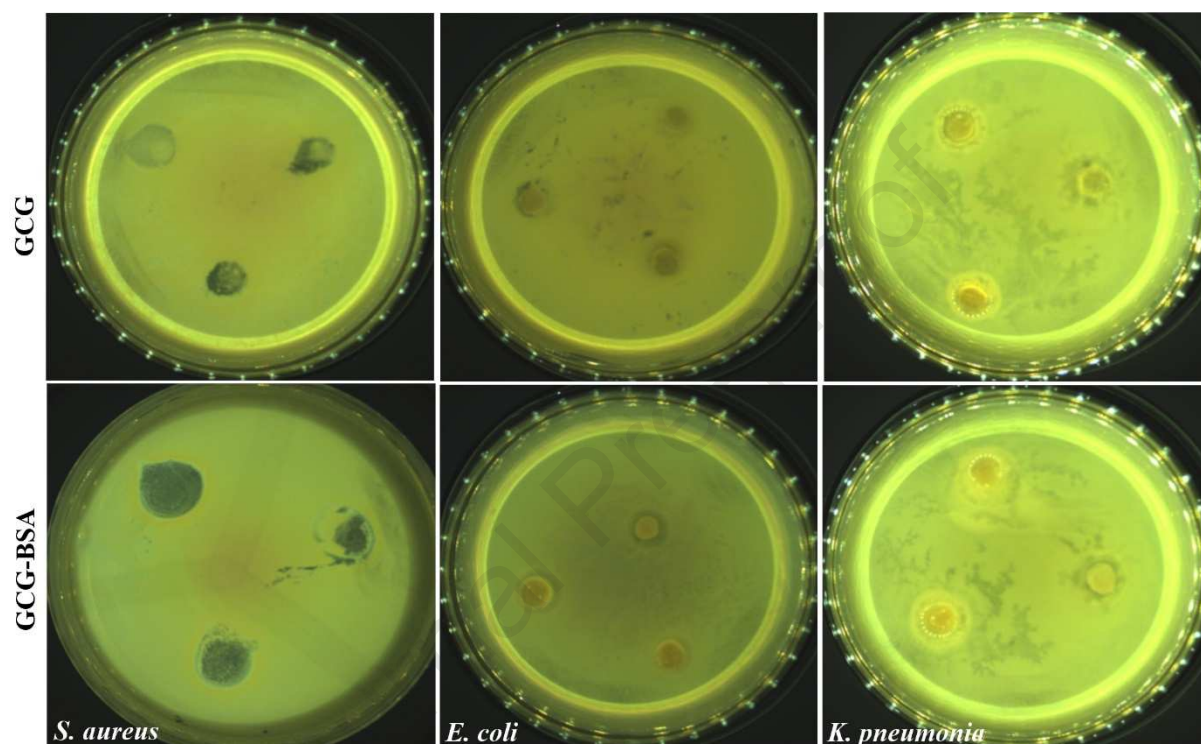
Kinetic models	Parameters	Samples	
		GC	GCB
Zero-order	K_1	0.862	0.708
	R^2	0.963	0.932
First-order	K_1	0.016	0.013
	M_o	78.59	74.35
	R^2	0.988	0.970
Higuchi	K_H	7.764	6.452
	R^2	0.999	0.998
Korsmeyer-Peppas	K_{KP}	11.108	5.441
	n	0.327	0.486
	R^2	0.999	0.982

553

554 3.5. Antibacterial activity of gel scaffolds

555 The antibacterial efficacy of the prepared GCB and drug-loaded GCB gel scaffolds were
 556 tested against three different bacteria of *S. aureus*, *E. coli* and *K. pneumonia*. Fig. 9 displays
 557 the bacteria resistance results of non-loaded and loaded gel scaffolds. According to
 558 observations, the non-loaded (GCB) gel scaffold showed resistance against all tested bacteria
 559 strains. This indicates that GCB possessed self-incorporated antibacterial properties. This
 560 self-antibacterial may be due to the presence of aldehyde groups in oxidized carrageenan
 561 gum. Similar results on antibacterial activity were achieved by Wang et al. in a study that
 562 reported the effects oxidized κ -carrageenan gum against *E. coli* and *S. aureus* bacteria [65].
 563 Another study by Zhu et al. investigated and reported the growth inhibition properties of
 564 periodate oxidized carrageenan gum against *E. coli*, *S. aureus*, *P. aeruginosa* and *L.*

565 *monocytogenes* bacteria [31]. In accordance with the results obtained from the present study,
 566 significant growth inhibition was achieved against *S. aureus* and *E. coli* while samples
 567 incubated in *K. pneumonia* medium showed resistance against the bacteria. Based on deduced
 568 results, the average growth inhibition zones in *S. aureus* and *E. coli* were determined as $4.1 \pm$
 569 0.20 mm and 2.30 ± 0.62 for GCB as compared to 5.4 ± 0.43 mm and 3.1 ± 0.88 mm for
 570 GCB-BSA gel scaffold, respectively.



571
 572 **Fig. 9.** Antibacterial properties of GCB and GCB-BSA gel scaffolds.

573 4. Conclusions

574 The present study investigates the preparation of injectable self-crosslinked
 575 gelatin/aldehyde carrageenan gum gel via the Schiff base nucleophilic reaction followed by
 576 interpenetration with bacterial cellulose. Based on results obtained, the prepared gels blended
 577 with bacterial cellulose (GCB1, GCB2 and GCB3) exhibited shorter gelation time, higher
 578 water uptake and lower *in vitro* degradation as compared to gels without BC (GC1 and GC2).
 579 In addition, characterization of the gels by various techniques revealed that the
 580 interpenetration of BC in the covalently crosslinked gelatin/aldehyde carrageenan gum
 581 network enhanced the tuneable properties. In comparison, GCB gels exhibited higher
 582 crystallinity and mechanical stability to GC. Also, GCB showed enhance resistance to shear
 583 deformation with variation in temperature and strain as well as possessed good shear thinning

584 properties. Furthermore, *in vitro* drug release studies demonstrated a sustain release was
585 achieved for GCB samples compared to GC. *In vitro* cytotoxicity assay on the non-loaded
586 and drug loaded GCB gel scaffolds showed significant and high cell viability. Based on
587 bacteria resistance, GCB demonstrated to possess self-antibacterial properties with growth
588 inhibition potency against gram positive and negative bacteria. Overall, the incorporation of
589 BC did not only improve the thermal gelation property of the crosslinked gel matrix, but
590 embedded improve mechanical stability as well as sustain drug release properties compared
591 to gels without BC. Based on good performance in mechanical stability, controlled release
592 and antibacterial activity, we believe that the prepared gel has potential opportunity to be
593 used as an injectable drug delivery system for wound healing and tissue regeneration
594 applications.

595 **Author contributions:**

596 This study conceptualization and methodology was performed by Fahanwi Asabuwa
597 Ngwabebhoh. Formal analysis was done by Fahanwi Asabuwa Ngwabebhoh, Rahul Patwa
598 and Oyunchimeg Zandraa. Investigation and data curation were carried out by Fahanwi
599 Asabuwa Ngwabebhoh, Rahul Patwa and Oyunchimeg Zandraa. Writing-original draft
600 preparation was done by Fahanwi Asabuwa Ngwabebhoh. The paper was reviewed, edited
601 and supervised by Nabanita Saha and Petr Saha.

602 **Declaration of Competing Interest**

603 The authors declare no conflict of interest.

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Declaration of interests

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

The authors declare the following financial interests/personal relationships which may be considered as potential competing interests:

None

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