

Frame-invariant formulation of novel Generalized Newtonian fluid constitutive equation for polymer melts

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ABSTRACT

In this work, the frame-invariant formulation of a new generalized Newtonian fluid (GNF) constitutive equation is proposed. Viscosity is given as a specific function of the second and third invariant of the strain rate tensor, and the second invariant of the objective velocity gradient. The GNF model was successfully tested using experimental data taken from the open literature for different high and low density polyethylene melts with varying amounts of long-chain branches utilizing steady-state shear, uniaxial and planar extensional viscosities plotted as a function of the strain rate.

The Generalized Newtonian Fluid (GNF) constitutive equation, in which the viscosity is allowed to vary with the second, II_D , and/or third, III_D , invariants of the strain rate tensor, D , [1-4] is widely used to model isotropic and incompressible complex fluid flows to improve convergence of solutions for very fast flows. The main disadvantage of the GNF model is its inability to realistically represent extensional rheology, especially in planar flows, where $III_D = 0$. To overcome this problem, Zatloukal [5-7] proposed a GNF model in which the viscosity is given as a specific function of II_D , III_D and the first invariant of the absolute value of the strain rate tensor, $I_{|D|}$, characterizing overall amount of stretching during the flow (where $|D| = \sqrt{D \cdot D}$). This model has been found to provide correct behaviour in describing steady-state shear and extensional viscosities for various polymer melts [5,7-9] and has been used successfully in modelling of the film blowing process [8, 10-11]. However, the use of the model in mixed shear and extensional flows is complicated because $I_{|D|}$ needs to be evaluated in a local, streamline oriented coordinate system to avoid multiple solutions and to keep its physical meaning [5]. In order to overcome this problem, a new modification of the GNF model is proposed here to provide a simple frame-invariant formulation of the model, maintaining its ability to describe the steady-state shear and extensional rheology of polymer melts.

To distinguish between different types of flows, we used an objective velocity gradient, \bar{L} , proposed by Yao [12-13], which is defined as

$$\bar{L} = D + \bar{W} \quad (1)$$

where D is the strain rate tensor and \bar{W} is the objective vorticity tensor defined as

$$D = \frac{1}{2}(L + L^T) \quad \text{and} \quad \bar{W} = \frac{1}{2}(L - L^T) - \Omega \quad (2)$$

Here, L represents the velocity gradient tensor and Ω is the rigid part of the body subtracted from the classical vorticity tensor, which is related to the rotation of the principal axes of D [12-13]. The rigid-body vorticity Ω is defined as

$$\frac{De_i}{Dt} = \Omega \cdot e_i \quad (3)$$

where $\frac{D}{Dt}$ is the substantial time derivative and e_i (for $i = 1, 2, 3$) are unit vectors along the principal axes of D . When at least two of the eigenvalues of D coincide, the corresponding components in Ω must be set equal to those in W , which makes the definition of Ω in Eq. 3 unique [12, 14]. Therefore, for steady-state flow, Ω disappears and \bar{L} and L becomes the same [12-13]. The GNF model, which was proposed in our previous work [5-7], is modified here by replacing the absolute value of the strain rate tensor, $|D|$, (defined as $\sqrt{D \cdot D}$) with the objective velocity gradient, \bar{L} . The modified GNF model has the following form

$$\tau = 2\eta \left(\Pi_{\bar{L}}, \Pi_D, \text{III}_D \right) D \quad (4)$$

where τ is the extra stress tensor and η is the viscosity, which is allowed to vary with the second $\Pi_{\bar{L}} = 2\text{tr}(\bar{L}^2)$ invariant of \bar{L} as well as with the second $\Pi_D = 2\text{tr}(D^2)$, and third, $\text{III}_D = \det(D)$, invariants of D according to Eq. 5

$$\eta(\Pi_{\bar{L}}, \Pi_D, \text{III}_D) = A^{1-f(\Pi_{\bar{L}}, \Pi_D, \text{III}_D)} \eta(\Pi_D)^{f(\Pi_{\bar{L}}, \Pi_D, \text{III}_D)} \quad (5)$$

where $\eta(\Pi_D)$ is given by Eq. 6 representing the well-known Carreau-Yasuda model and $f(\Pi_{\bar{L}}, \Pi_D, \text{III}_D)$ is given by Eq. 7

$$\eta(\Pi_D) = \eta_\infty + \frac{\eta_0 - \eta_\infty}{\left[1 + (\lambda_1 \sqrt{\Pi_D})^a \right]^{\left(\frac{1-n}{a} \right)}} \quad (6)$$

$$f(\Pi_L, \Pi_D, \Pi_D) = \left\{ \tanh \left[\lambda_2 \left(1 + \frac{1}{12\sqrt{3}} \right)^{-\psi} \left(1 + \frac{\Pi_D}{\Pi_D^{3/2}} \right)^\psi \sqrt{\Pi_L} + \beta \right] \frac{1}{\tanh(\beta)} \right\}^\zeta \quad (7)$$

where A , η_0 , η_∞ , λ_1 , a , n , λ_2 , ψ , β , ζ are 10 adjustable parameters.

The model utilizes 5 parameters (η_0 , η_∞ , λ_1 , a , n) identifiable from shear viscosity, 4 parameters (A , λ_2 , β , ζ) identifiable from uniaxial extensional viscosity and 1 parameter (ψ), which can be determined from the planar extensional viscosity. In a simple shear flow, the function $f(\Pi_L, \Pi_D, \Pi_D)$ defined by Eq. 7 is 1 because $\sqrt{\Pi_L} = 0$. On the other hand, in extensional flows, the term $\sqrt{\Pi_L}$ becomes nonzero (equal to $\sqrt{3}\dot{\epsilon}$ for uniaxial flow, $2\sqrt{3}\dot{\epsilon}$ for equibiaxial flow, and $2\dot{\epsilon}$ for planar flow) and as a result, the function $f(\Pi_L, \Pi_D, \Pi_D)$ starts to deviate from 1. The adjusted value of the term

$\left(1 + \frac{1}{12\sqrt{3}} \right)^{-\psi} \left(1 + \frac{\Pi_D}{\Pi_D^{3/2}} \right)^\psi$ via the parameter ψ enables independent control of planar and equibiaxial extensional viscosity, while the shear and uniaxial extensional viscosity do not change as in the original model [5-7]. The key difference between the modified GNF model and its original version is the $\sqrt{\Pi_L}$

term that appears in the function f , instead of the original $\frac{\sqrt[3]{4|\Pi_D| + I_{|D|}}}{3}$ term. The main invariants for D and \bar{L} are given in Tables 1 and 2 for simple and mixed flows. It can be seen that for pure shear and extensional flow Π_L becomes zero and non-zero, respectively, and it depends only on the extensional strain rate, regardless of the type of mixed flow. Π_L can therefore be considered as a variable characterizing the intensity of stretching during flow.

The modified GNF model provides analytical expressions for shear viscosity, η_s , as well as uniaxial, η_{EU} , biaxial, η_{EB} , and planar, η_{EP} , extensional viscosities, which are summarized below.

$$\eta_s = \eta_\infty + \frac{\eta_0 - \eta_\infty}{\left[1 + (\lambda_1 \dot{\gamma})^a\right]^{\frac{1-n}{a}}} \quad (8)$$

$$\eta_{E,U} = 3 * A^{1 - \left[\frac{\tanh(\lambda_2 \sqrt{3} \dot{\epsilon} + \beta)}{\tanh(\beta)} \right]^\zeta} \left\{ \eta_\infty + \frac{\eta_0 - \eta_\infty}{\left[1 + (\lambda_1 \sqrt{3} \dot{\epsilon})^a\right]^{\frac{1-n}{a}}} \right\}^{\left[\frac{\tanh(\lambda_2 \sqrt{3} \dot{\epsilon} + \beta)}{\tanh(\beta)} \right]^\zeta} \quad (9)$$

$$\eta_{E,B} = 6 * A^{1 - \left\{ \frac{\tanh \left[\lambda_2 \left(1 - \frac{2}{1+12\sqrt{3}} \right)^\nu 2\sqrt{3} \dot{\epsilon} + \beta \right]}{\tanh(\beta)} \right\}^\zeta} \left\{ \eta_\infty + \frac{\eta_0 - \eta_\infty}{\left[1 + (\lambda_1 2\sqrt{3} \dot{\epsilon})^a\right]^{\frac{1-n}{a}}} \right\}^{\left\{ \frac{\tanh \left[\lambda_2 \left(1 - \frac{2}{1+12\sqrt{3}} \right)^\nu 2\sqrt{3} \dot{\epsilon} + \beta \right]}{\tanh(\beta)} \right\}^\zeta} \quad (10)$$

$$\eta_{E,P} = 4 * A^{1 - \left\{ \frac{\tanh \left[\lambda_2 \left(1 - \frac{1}{1+12\sqrt{3}} \right)^\nu 2\dot{\epsilon} + \beta \right]}{\tanh(\beta)} \right\}^\zeta} \left\{ \eta_\infty + \frac{\eta_0 - \eta_\infty}{\left[1 + (\lambda_1 2\dot{\epsilon})^a\right]^{\frac{1-n}{a}}} \right\}^{\left\{ \frac{\tanh \left[\lambda_2 \left(1 - \frac{1}{1+12\sqrt{3}} \right)^\nu 2\dot{\epsilon} + \beta \right]}{\tanh(\beta)} \right\}^\zeta} \quad (11)$$

In order to test the modified GNF model, steady-state shear, uniaxial and planar extensional viscosities plotted as a function of the strain rate were used for highly branched low-density polyethylenes, LDPEs as well as metallocene high-density (mHDPE) and linear low-density (mLLDPE) polyethylenes with varying amounts of long-chain branches. Experimental data were taken from [15] and the fitting procedure was as follows. First, the shear viscosity was fitted by the Carreau-Yasuda function (Eq. 8) to determine η_0 , η_∞ , λ_1 , a , n parameters. Second, A , λ_2 , β , ζ model parameters were identified on uniaxial extensional viscosity data via Eq. 9 (keeping η_0 , η_∞ , λ_1 , a , n parameters unchanged). Third, the remaining ψ parameter was determined by fitting the planar extensional viscosity data by Eq. 11 (keeping all other parameters fixed). In the final stage, shear, uniaxial and planar extensional viscosities were fitted simultaneously via Eqs. 8-9, 11 using the parameter values determined in previous three steps as initial values. Note that due to the absence of a secondary high strain rate plateau, the parameters η_∞ and A were fixed at 0 Pa.s and 10^{-8} Pa.s, respectively. A comparison of model fits and measured data is shown

in Figures 1-2 and the corresponding model parameters are listed in Tables 3-4. It can be seen that the modified GNF model has a very good ability to describe measured data for all polymer samples tested. The physical role of the 5 extension related parameters is as follows: λ_2 , ζ and β dominantly controls the uniaxial extensional strain hardening (their values have tendency to increase with LCB, see Table 4), ψ allows independent control of steady planar/equibiaxial extensional viscosity (its increase reduces strain hardening and promotes strain thinning) and A parameter is related to the high-extension rate plateau value of the extensional viscosity. The model has the ability to describe lower $\eta_{E,P}$ at high strain rates compared to $\eta_{E,U}$ for highly branched LDPEs (see Figure 1) as well as a gradual decrease in $\eta_{E,P}$ relative to $\eta_{E,U}$ at high strain rates with increased level of long chain branching for mLLDPE and mHDPE polymer samples through an increased parameter ψ (see Figure 2 and Table 4). It is important to note that various widely used constitutive equations (including the molecular-based Pom-Pom model [15], see Figure S1 in supplementary material) are not able to handle these differences because they predict steady-state uniaxial and planar extensional viscosities to become virtually identical at high extensional strain rates [16], which is not realistic for the studied polymers. This suggests that the use of the modified GNF model in modeling of polymer processing can significantly help with material, equipment design and process optimization, and can provide explanations for flow phenomena that are not yet fully understood (for example vortex formation [17-18], neck-in phenomenon [16, 19-21] or instabilities in the melt blowing process [22-24], etc.).

See supplementary material for additional information, which includes a comparison between the modified GNF model fits and corresponding experimental rheological data for well-characterized branched IUPAC A LDPE and linear HDPE Liten polymer melts, which are available in a wide range of strain rates.

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The data that supports the findings of this study are available within the article.

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Table 1 Three principal invariants of the deformation rate tensor D and the objective velocity gradient \bar{L} .

| | Simple shear flow | Uniaxial extensional flow | Equibiaxial flow | Planar flow |
|-------------------------|---|--|--|--|
| D | $\begin{pmatrix} 0 & \dot{\gamma}/2 & 0 \\ \dot{\gamma}/2 & 0 & 0 \\ 0 & 0 & 0 \end{pmatrix}$ | $\begin{pmatrix} \dot{\epsilon} & 0 & 0 \\ 0 & -\dot{\epsilon}/2 & 0 \\ 0 & 0 & -\dot{\epsilon}/2 \end{pmatrix}$ | $\begin{pmatrix} \dot{\epsilon} & 0 & 0 \\ 0 & \dot{\epsilon} & 0 \\ 0 & 0 & -2\dot{\epsilon} \end{pmatrix}$ | $\begin{pmatrix} \dot{\epsilon} & 0 & 0 \\ 0 & 0 & 0 \\ 0 & 0 & -\dot{\epsilon} \end{pmatrix}$ |
| $\bar{L} = L$ | $\begin{pmatrix} 0 & \dot{\gamma} & 0 \\ 0 & 0 & 0 \\ 0 & 0 & 0 \end{pmatrix}$ | $\begin{pmatrix} \dot{\epsilon} & 0 & 0 \\ 0 & -\dot{\epsilon}/2 & 0 \\ 0 & 0 & -\dot{\epsilon}/2 \end{pmatrix}$ | $\begin{pmatrix} \dot{\epsilon} & 0 & 0 \\ 0 & \dot{\epsilon} & 0 \\ 0 & 0 & -2\dot{\epsilon} \end{pmatrix}$ | $\begin{pmatrix} \dot{\epsilon} & 0 & 0 \\ 0 & 0 & 0 \\ 0 & 0 & -\dot{\epsilon} \end{pmatrix}$ |
| $I_D = I_{\bar{L}}$ | 0 | 0 | 0 | 0 |
| II_D | $\dot{\gamma}^2$ | $3\dot{\epsilon}^2$ | $12\dot{\epsilon}^2$ | $4\dot{\epsilon}^2$ |
| $II_{\bar{L}}$ | 0 | $3\dot{\epsilon}^2$ | $12\dot{\epsilon}^2$ | $4\dot{\epsilon}^2$ |
| $III_D = III_{\bar{L}}$ | 0 | $\frac{1}{4}\dot{\epsilon}^3$ | $-2\dot{\epsilon}^3$ | 0 |

Here, $\dot{\gamma}$ and $\dot{\epsilon}$ represent shear and extensional strain rates, respectively.

Table 2 Three principal invariants of the deformation rate tensor D and the objective velocity gradient \bar{L} in the mixed flows.

| | Shear flow + uniaxial extensional flow | Shear flow + equibiaxial flow | Shear flow + planar flow |
|---------------------|--|--|--|
| D | $\begin{pmatrix} \dot{\epsilon} & \dot{\gamma}/2 & 0 \\ \dot{\gamma}/2 & -\dot{\epsilon}/2 & 0 \\ 0 & 0 & -\dot{\epsilon}/2 \end{pmatrix}$ | $\begin{pmatrix} \dot{\epsilon} & \dot{\gamma}/2 & 0 \\ \dot{\gamma}/2 & \dot{\epsilon} & 0 \\ 0 & 0 & -2\dot{\epsilon} \end{pmatrix}$ | $\begin{pmatrix} \dot{\epsilon} & \dot{\gamma}/2 & 0 \\ \dot{\gamma}/2 & 0 & 0 \\ 0 & 0 & -\dot{\epsilon} \end{pmatrix}$ |
| $\bar{L} = L$ | $\begin{pmatrix} \dot{\epsilon} & \dot{\gamma} & 0 \\ 0 & -\dot{\epsilon}/2 & 0 \\ 0 & 0 & -\dot{\epsilon}/2 \end{pmatrix}$ | $\begin{pmatrix} \dot{\epsilon} & \dot{\gamma} & 0 \\ 0 & \dot{\epsilon} & 0 \\ 0 & 0 & -2\dot{\epsilon} \end{pmatrix}$ | $\begin{pmatrix} \dot{\epsilon} & \dot{\gamma} & 0 \\ 0 & 0 & 0 \\ 0 & 0 & -\dot{\epsilon} \end{pmatrix}$ |
| $I_D = I_{\bar{L}}$ | 0 | 0 | 0 |
| II_D | $3\dot{\epsilon}^2 + \dot{\gamma}^2$ | $12\dot{\epsilon}^2 + \dot{\gamma}^2$ | $4\dot{\epsilon}^2 + \dot{\gamma}^2$ |
| $II_{\bar{L}}$ | $3\dot{\epsilon}^2$ | $12\dot{\epsilon}^2$ | $4\dot{\epsilon}^2$ |
| III_D | $\frac{1}{4}\dot{\epsilon}^3 + \frac{1}{8}\dot{\gamma}^2\dot{\epsilon}$ | $-2\dot{\epsilon}^3 + \frac{1}{2}\dot{\gamma}^2\dot{\epsilon}$ | $\frac{\dot{\gamma}^2\dot{\epsilon}}{4}$ |
| $III_{\bar{L}}$ | $\frac{1}{4}\dot{\epsilon}^3$ | $-2\dot{\epsilon}^3$ | 0 |

Here, $\dot{\gamma}$ and $\dot{\epsilon}$ represent shear and extensional strain rates, respectively.

Table 3 Parameters of the GNF model for LDPE1 (T=140°C), LDPE2 (T=150°C) and LDPE3 (T=160°C). $A=10^{-8}$ Pa.s for all samples.

| Sample name | η_0 (Pa.s) | λ_1 (s) | a (-) | n (-) | λ_2 (s) | β (-) | ζ (-) | ψ (-) |
|------------------------------------|--------------------|--------------------|------------|------------|--------------------------|------------------------|-------------------------|---------------|
| LDPE1 (high strain hardening) | 3266.4 | 0.0933 | 0.7696 | 10^{-6} | $507.9532 \cdot 10^{-4}$ | $1.7149 \cdot 10^{-2}$ | $24.7119 \cdot 10^{-3}$ | 35.5205 |
| LDPE2 (medium strain hardening) | 50829.6 | 8.4844 | 0.4904 | 10^{-6} | $1.7432 \cdot 10^{-4}$ | $9.3813 \cdot 10^{-8}$ | $16.0381 \cdot 10^{-3}$ | 35.4614 |
| LDPE3 (low strain hardening) | 413830.6 | 23.0778 | 0.5124 | 0.1823 | $12.1625 \cdot 10^{-4}$ | $1.1320 \cdot 10^{-8}$ | $7.4815 \cdot 10^{-3}$ | 77.2106 |

Table 4 Parameters of the GNF model for metallocene high-density (mHDPE) and linear low-density (mLLDPE) polyethylenes with varying amounts of long-chain branches (LCBs) at 155°C. $A=10^{-8}$ Pa.s for all samples.

| Sample name | η_0 (Pa.s) | λ_1 (s) | a (-) | n (-) | λ_2 (s) | β (-) | ζ (-) | ψ (-) |
|-------------------|--------------------|--------------------|------------|------------|--------------------------|------------------------|------------------------|---------------|
| LCB-low mHDPE1 | 9687.5 | 2.2177 | 0.7571 | 0.6510 | $501.5086 \cdot 10^{-6}$ | $9.2969 \cdot 10^{-7}$ | $2.3349 \cdot 10^{-3}$ | 32.1062 |
| LCB-medium mHDPE2 | 28481.6 | 9.3122 | 0.7738 | 0.5899 | 4.5913 | $7.8399 \cdot 10^{-4}$ | $3.3020 \cdot 10^{-3}$ | 73.2949 |
| LCB-high mHDPE3 | 47948.3 | 2.6126 | 0.6179 | 0.3418 | 1.3166 | $1.1719 \cdot 10^{-3}$ | $5.1234 \cdot 10^{-3}$ | 127.2642 |
| Linear mLLDPE1 | 11654.9 | 1.1682 | 1.5646 | 0.8291 | $2.5161 \cdot 10^{-7}$ | $1.0314 \cdot 10^{-7}$ | $1.9825 \cdot 10^{-3}$ | 0 |
| LCB-low mLLDPE2 | 39440.6 | 1.5258 | 0.5333 | 0.4234 | $3.1024 \cdot 10^{-4}$ | $5.8652 \cdot 10^{-7}$ | $3.8623 \cdot 10^{-3}$ | 15.0290 |
| LCB-high LLDPE3 | 32145.8 | 0.7586 | 0.5717 | 0.1698 | $8.4339 \cdot 10^{-4}$ | $6.9952 \cdot 10^{-6}$ | $8.0257 \cdot 10^{-3}$ | 24.2860 |

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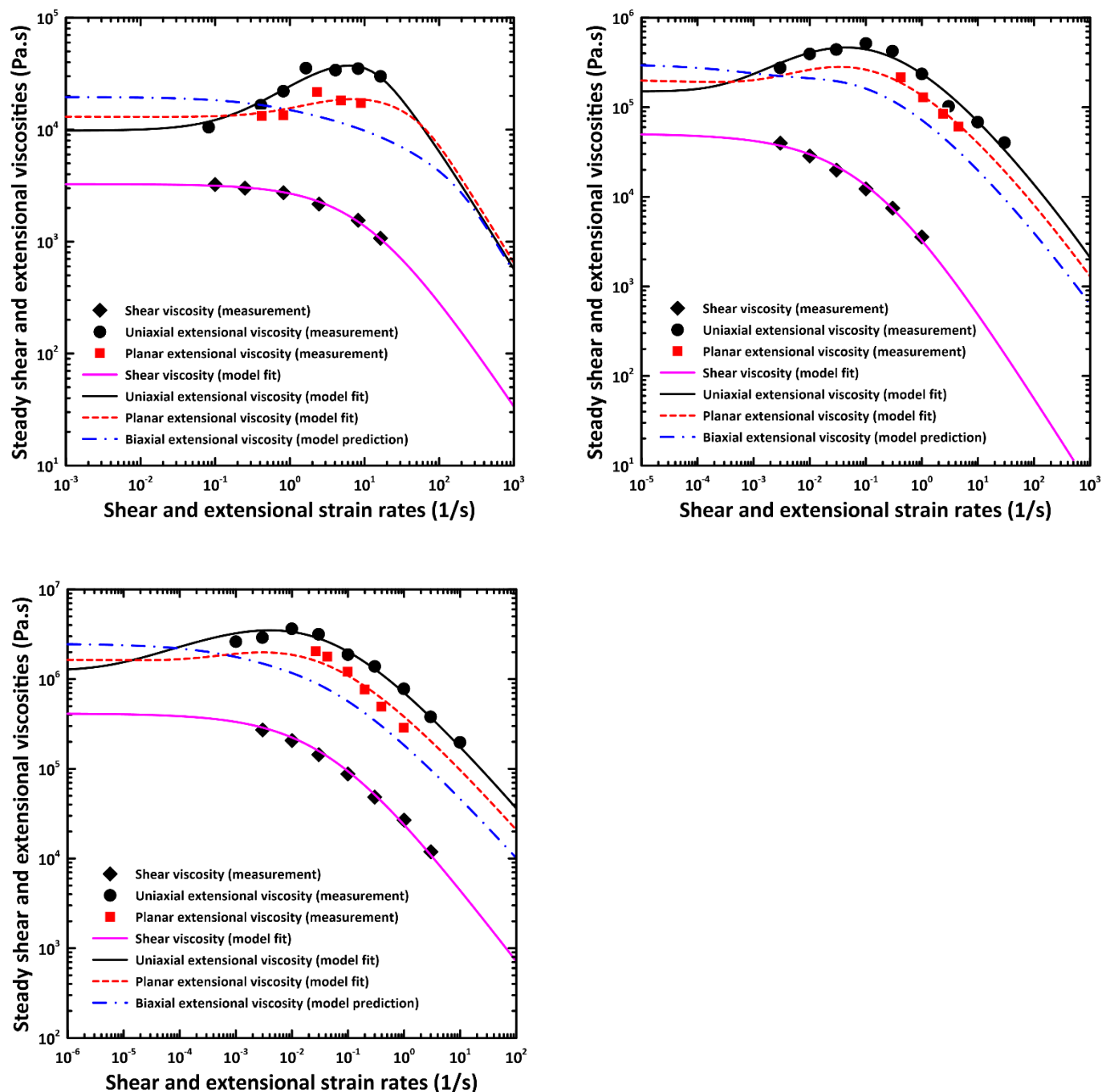


Figure 1 Comparison between the GNF model fits/prediction and the measured strain rate dependent shear and extensional viscosities for branched LDPE1 (left, top) LDPE2 (right, top) and LDPE3 (left, bottom). The measured data are taken from [15].

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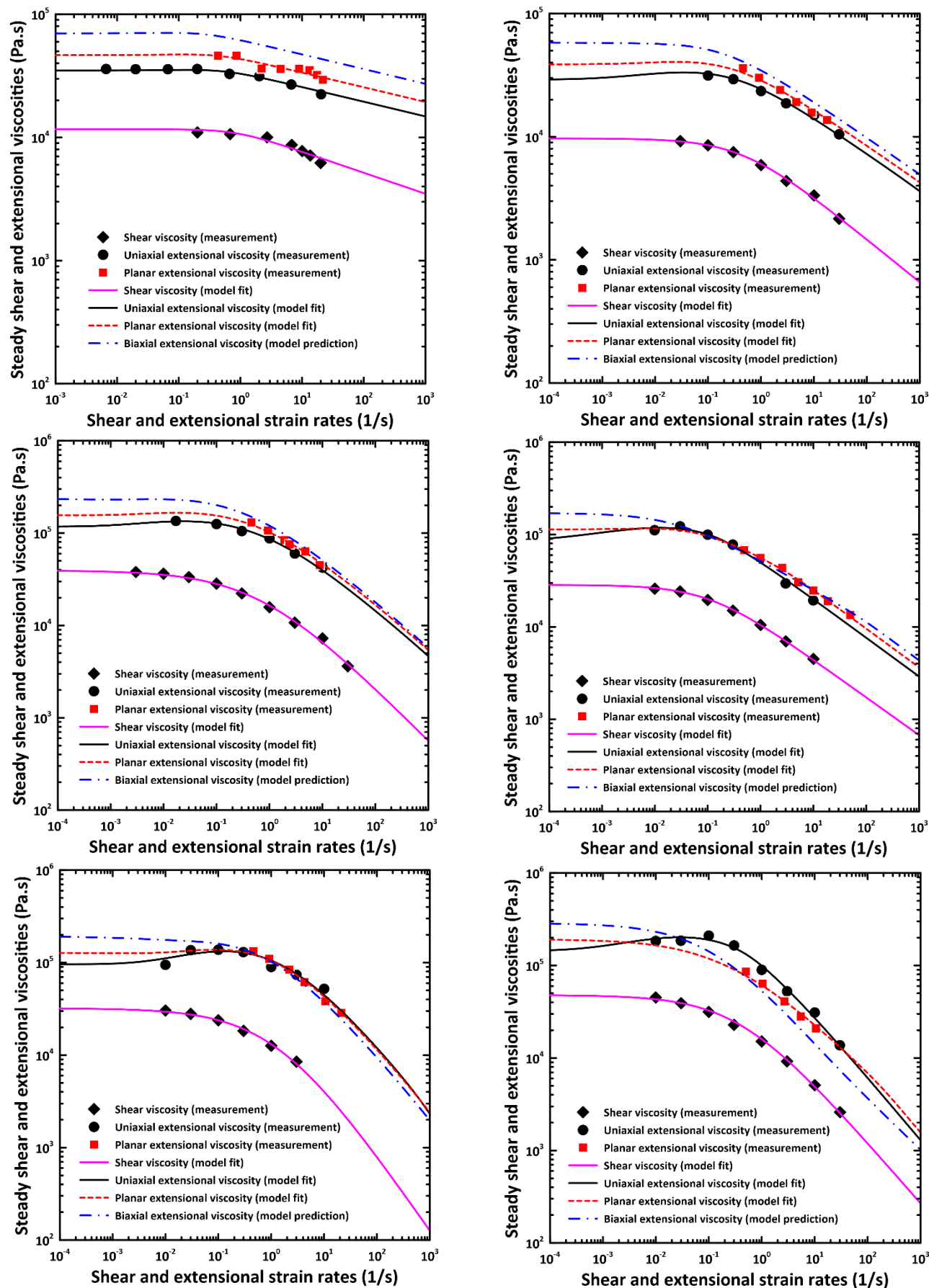


Figure 2 Comparison between the GNF model fits/predictions and the measured strain rate dependent shear and extensional viscosities for linear mLLDPE1 (left, top), slightly branched mLLDPE2 (left, in the middle), highly branched mLLDPE3 (left, bottom), slightly branched mHDPE1 (right, top), mHDPE2 medium branched (right, in the middle) and highly branched mHDPE3 (right, bottom). The measured data are taken from [15].

